

Realizing enhanced energy storage and hardness performances in 0.90NaNbO₃–0.10Bi(Zn_{0.5}Sn_{0.5})O₃ ceramics

Xiaoyan DONG^a, Xu LI^b, Hongyun CHEN^a, Qinpeng DONG^a, Jiaming WANG^a, Xiang WANG^a, Yue PAN^a, Xiuli CHEN^{a,*},
Huanfu ZHOU^a

^a*Collaborative Innovation Center for Exploration of Hidden Nonferrous Metal Deposits and Development of New Materials in Guangxi, Key Laboratory of Nonferrous Materials and New Processing Technology, Ministry of Education, School of Materials Science and Engineering, Guilin University of Technology, Guilin 541004, China*

^b*College of Materials Science and Engineering, Sichuan University, Chengdu 610064, China*

* Corresponding author.

E-mail: cxlnwpu@163.com

1 Experimental procedure

Based on stoichiometric proportions, the (1-*x*)NaNbO₃-*x*Bi(Zn_{0.5}Sn_{0.5})O₃ (in brief as (1-*x*)NN-*x*BZS, *x* = 0.05, 0.10, 0.15, and 0.20) ceramics were composited by a traditional solid-state reaction technique. Na₂CO₃ (≥ 99%), Nb₂O₅ (≥ 99%), Bi₂O₃ (≥ 99%), ZnO (≥ 99%), and SnO₂ (≥ 99.5%) were chosen as original materials. These natural materials were weighed and wet-milled in alcohol and ZrO₂ for 4 h. Then, the resulting slurry fluid was dried and calcined at 800 °C for 4 h in air. After calcining, the powders were wet-milled for 4 h, and then pressed into pellets with 8 mm in diameter and 1 mm in thickness using 5 wt% polyvinyl alcohol (PVA) as the binder. Finally, the pellets were sintered at 1200–1300 °C for 2 h after removing PVA.

The phase structures were tested by an X-ray diffractometer (XRD) (Cu Kα1, 1.54059 Å, Model X'Pert PRO, PANalytical, the Netherlands). Free surface of the samples was analyzed by a field emission scanning electron microscope (SEM) (FE–

SEM, Model S4800, Hitachi, Japan) equipped with an energy-dispersive X-ray spectroscopy (EDS). The selected area electron diffraction (SAED) and high-resolution transmission electron microscopy (HRTEM) imaging were acquired by JEM-2100F. The temperature-dependence of ϵ_r and $\tan\delta$ for x BZS ceramics was measured at a frequency of 10 kHz by a precision impedance analyzer (4294A, Hewlett–Packard, Palo Alto, USA). Ferroelectric hysteresis (P – E) loops were measured with the ferroelectric analyzer (RT66, Radiant Technologies, USA), and the thickness and electrode area of all the test samples were 0.10–0.12 mm and 0.0314 cm², respectively. Unipolar hysteresis loops were carried out based on the half-period triangular waveform. PFM amplitude and phase images were obtained by a piezoresponse force microscope (PFM, MFP-3D, USA), which was using Pt/Ir-coated conductive tips (Nanosensors, Neuchatel, Switzerland) with an AC tip voltage of 3 V and out-of-plane mode. The charge and discharge characteristics of ceramics were examined using a dielectric charge and discharge test system (CFD-003, TG Technology, Shanghai, China). The nanoindentation testing system (iNano, KLA, USA) conducted by a Berkovich indenter and a Vickers diamond indenter (HMV-G31DT, Shimadzu, Japan) was used to measure the hardness of ceramics with a polished surface. The indentation load was 4.903 N, together with a holding time of 15 s. The corresponding 3D morphologies were detected via the Bruker Countour GT K 3D.

2 Results and discussion

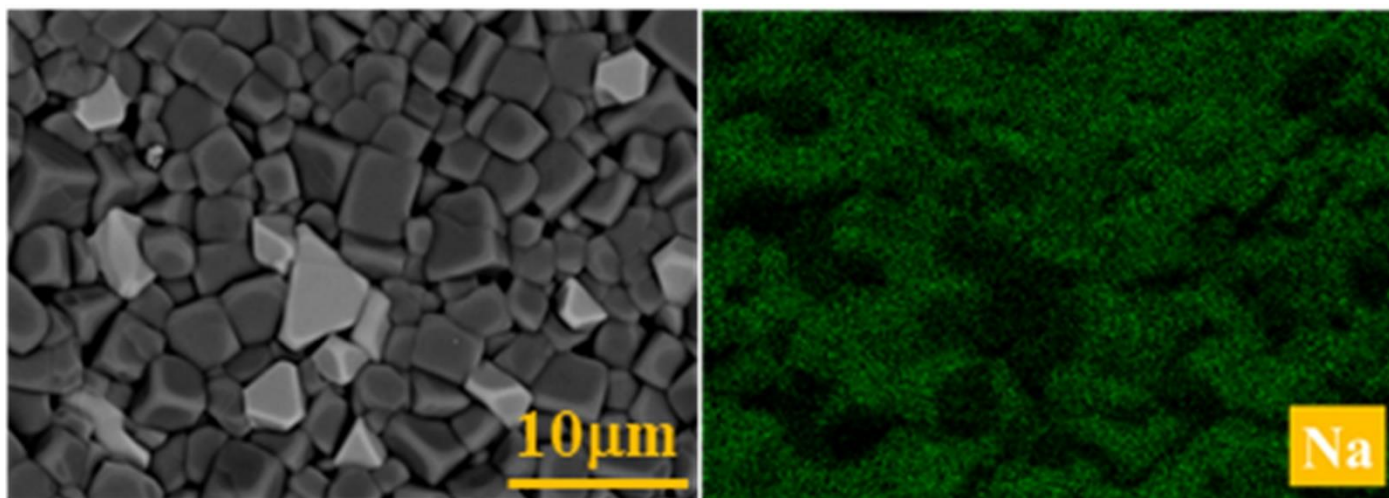


Fig. S1 Elemental mapping of 0.20BZS ceramics.

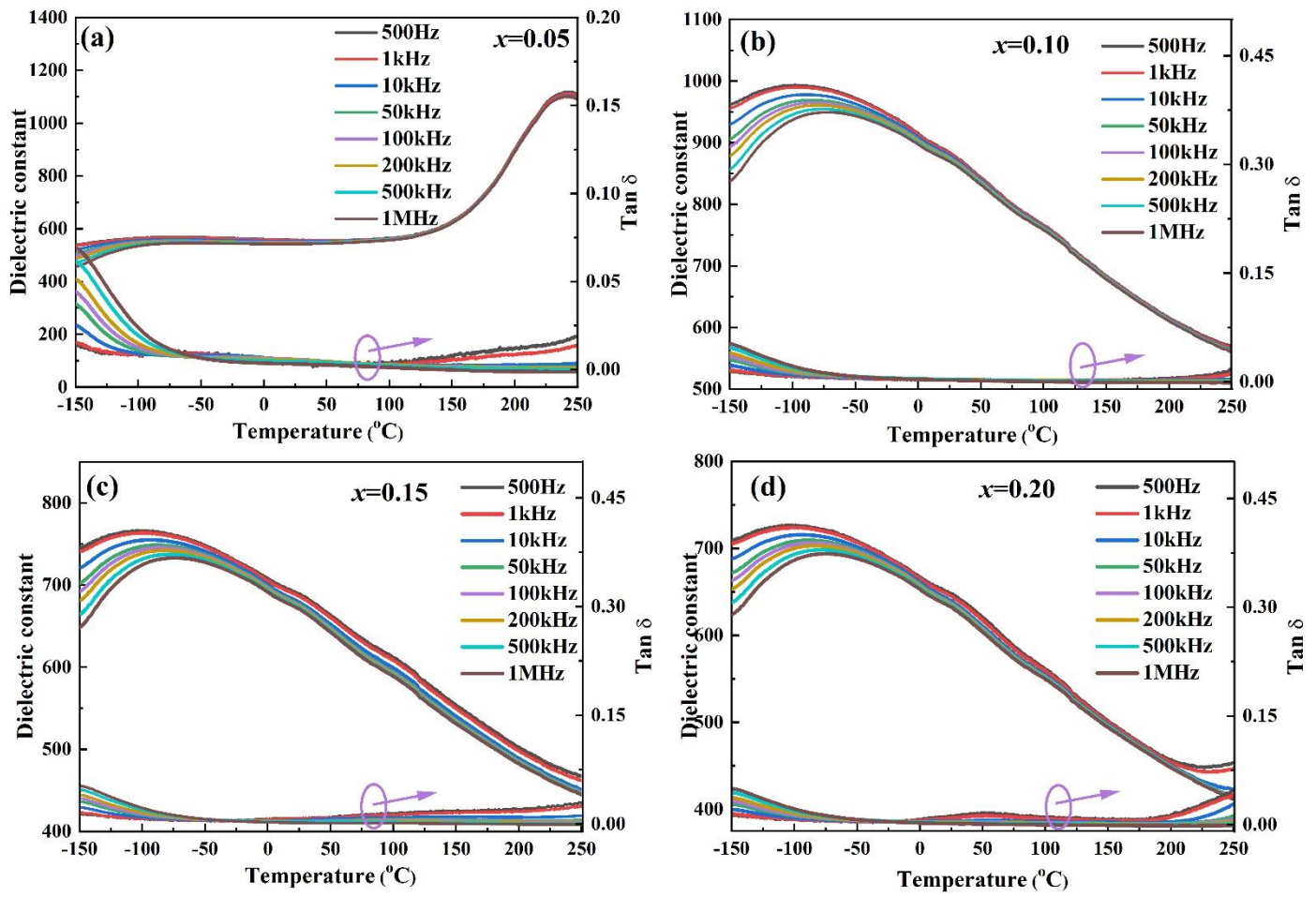


Fig. S2 Temperature dependences of dielectric constant and dielectric loss for $(1-x)\text{NN}-x\text{BZS}$ ceramics.

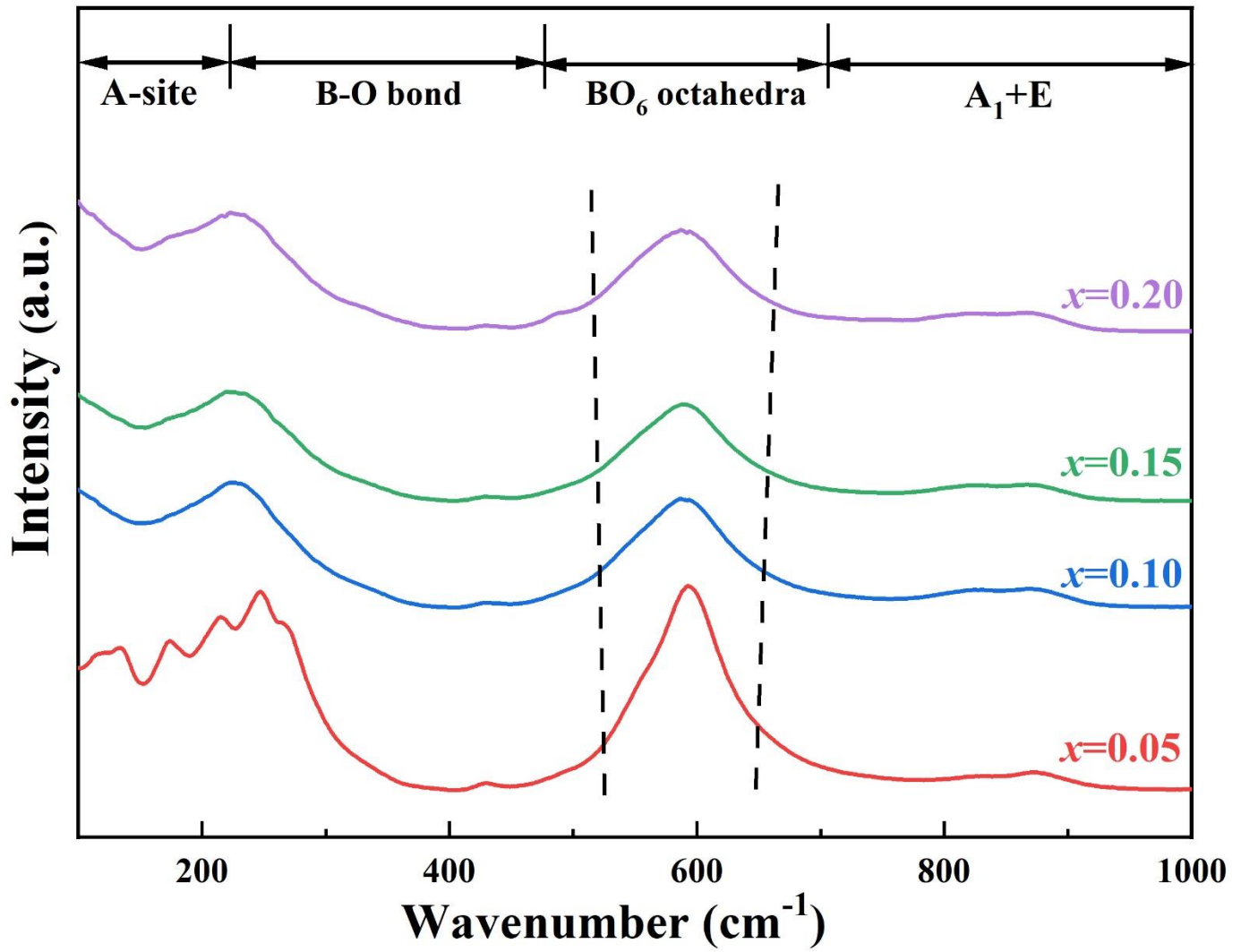


Fig. S3 Raman spectra of x BZS ceramics at room temperature.

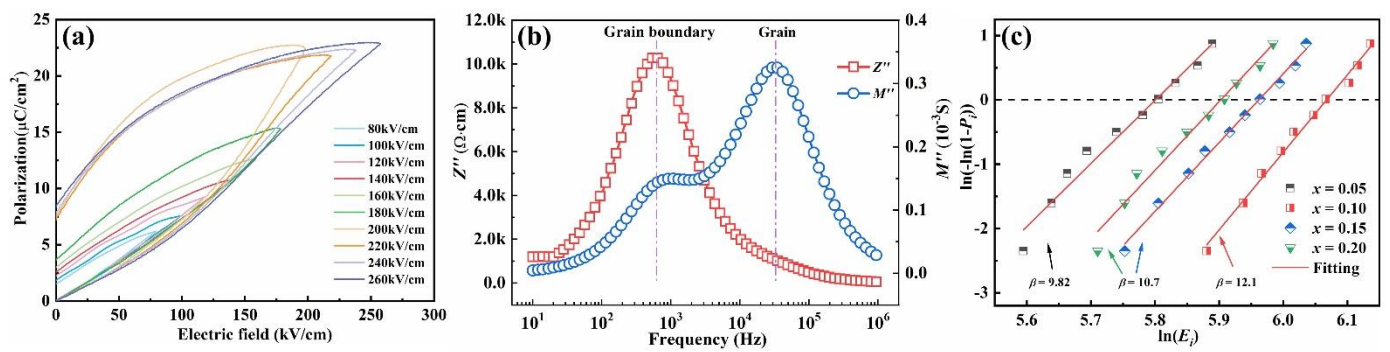


Fig. S4 (a) Unipolar P - E loops at various electrical fields of 0.05BZS ceramics. (b) Imaginary part of impedance (Z'') and modulus (M'') in 0.10BZS ceramics as a function of frequency measured in 500 °C. (c) Weibull distribution function of dielectric breakdown strength for x BZS ceramics.

Table S1 Mechanical properties of x BZS ceramics ($x = 0.00$ and 0.10) derived from Vickers indentation test.

Component	P_m (N)	d (μm)	H (GPa)
0.00	4.903	44.10	4.68
0.10	4.903	40.68	5.49