

Supporting Information

The Central Role of Tilted Anisotropy for Field-Free Spin-Orbit Torque Switching of Perpendicular Magnetization

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1. Wedge effect in films prepared on wedge-shape holder

The thickness difference across the films prepared by oblique angle deposition was characterized using an atomic force microscope (AFM). **Fig. S1(a)** demonstrates the height profile of a step across the SiO₂ substrate and the obliquely-deposited Pt film, which was prepared on a 40°-holder with a growth time of 300 s and sputtering power of 30 W. The average thickness gradient along the y direction is given as $\Delta h/\Delta y = 0.06 \text{ nm}/\mu\text{m}$ with average height of 13.1 nm. Therefore, in the 5 nm thick underlayer Pt used in the $\theta_{\text{dep}} = 40^\circ$ -based devices, the thickness difference across the 5 μm -wide Hall channel in the transverse direction approximately is $(0.06 \text{ nm}/\mu\text{m}) \times (5 \mu\text{m}) \times \frac{5 \text{ nm}}{13.1 \text{ nm}} = 0.1 \text{ nm}$. And for the case of uniformly deposited Co layer prepared on a rotating 40°-holder, its wedge effect is investigated as well. The height profile of a step prepared with a growth time of 600 s and sputtering power of 30 W is shown in **Fig. S1(b)**, which gives $\Delta h/\Delta y = -0.17 \text{ nm}/\mu\text{m}$ and average height of 12.3 nm. For the 0.5 nm Co layer in the $\theta_{\text{dep}} = 40^\circ$ -based devices, the estimated transverse thickness difference across the 5 μm -wide Hall channel is accordingly to be $(0.17 \text{ nm}/\mu\text{m}) \times (5 \mu\text{m}) \times \frac{0.5 \text{ nm}}{12.3 \text{ nm}} = 0.03 \text{ nm}$. Both cases imply the wedge effect are not substantial as using a wedge-shape sample holder to obliquely depositing Pt layer and uniformly depositing Co layer.

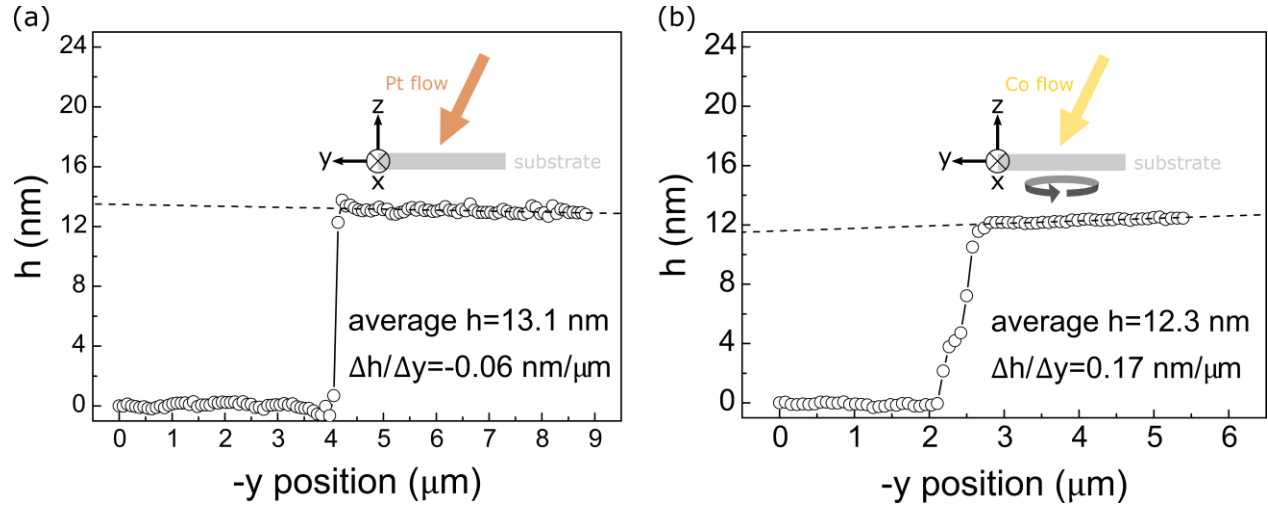


Fig. S1 Height profiles of steps crossing SiO₂ substrate and the deposited films, which were characterized by an atomic force microscope. a An obliquely deposited Pt film and **b** an uniformly deposited Co film, both are prepared on a 40°-holder. The orange and yellow arrows indicate the Pt and Co atom flow coming from the sputtering targets, respectively. The deposition angles are illustrated rather depicting the exact direction.

2. Damping-like SOT efficiency of obliquely-deposited Pt/Co/Pt layers

The damping-like SOT efficiencies (ξ_{DL}) of the uniformly and obliquely-deposited Pt/Co/Pt layers were characterized according to the framework of anomalous Hall effect (AHE) hysteresis loop shift measurement¹. The perpendicular effective SOT field generated per applied current (H_z^{eff}/I) as functions of external in-plane field (H_x) are summarized in **Fig. S2**, where the applied current is ranging from -2.5 mA to +2.5 mA. In this small range of applied current, the estimated ξ_{DL} are given as 0.09 for $\theta_{dep} = 0^\circ$ and 0.14 for $\theta_{dep} = 40^\circ$, which are consistent with the previously reported value² of efficiency of Pt.

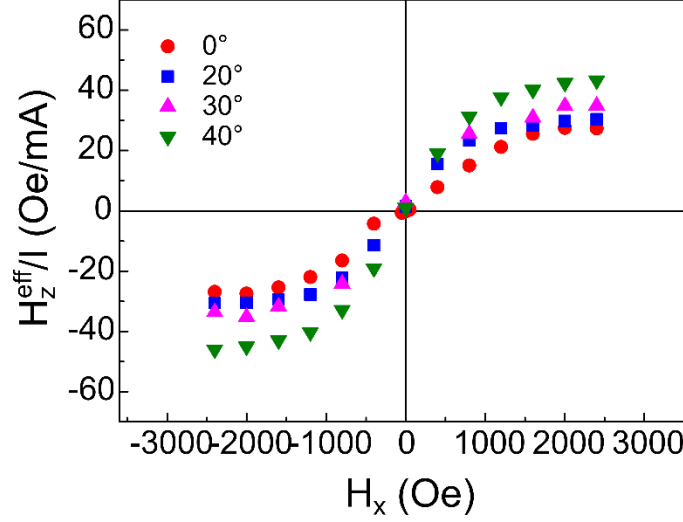


Fig. S2 DL-SOT characterization. Perpendicular SOT effective field per current (H_z^{eff}/I) varying with different longitudinal magnetic field (H_x). The data are characterized from the Pt/Co/Pt layers prepared by different deposition angle (0° to 40°).

3. Out-of-plane anisotropy field of obliquely-deposited Pt/Co/Pt layers

The value of out-of-plane anisotropy field (H_k) of the Pt/Co/Pt layers was determined through the response of out-of-plane magnetization (m_z) to the external in-plane magnetic field³. **Fig. S3** shows the m_z reduction behavior of devices with and without tilted anisotropy, which are respectively prepared by $\theta_{\text{dep}} = 0^\circ$ and 40° . In **Fig. S3(a)**, the device without tilted anisotropy shows a symmetric m_z reduction behavior as applying transverse external magnetic field (H_y), which is quite different from the response of the device with tilted anisotropy shown in **Fig. S3(b)**, that an out-of-plane magnetization switching happens as applying a sufficiently large H_y . Such switching is consistent with the scenario of laterally tilted anisotropy that the H_y projected onto the easy axis is non-zero for the case with tilted anisotropy. Additionally, the value of H_k can be

extracted through $1/H_k = \sqrt{|\partial^2 m_z / \partial H_y^2|}$, giving H_k approximately 5824 Oe and 7432 Oe for the devices with and without tilted anisotropy, respectively. This result implies that the presence of tilted anisotropy would slightly reduce the out-of-plane anisotropy.

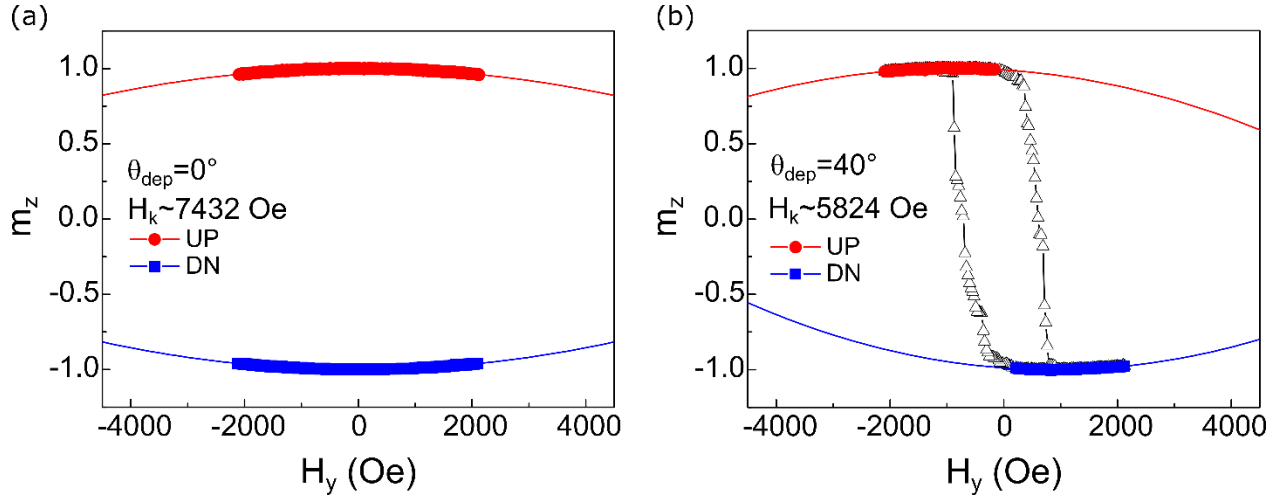


Fig. S3 Anisotropy field characterization. Response of out-of-plane magnetization (m_z) to external transverse magnetic field (H_y) of obliquely-deposited Pt/Co/Pt layers. a, b Devices without/with tilted anisotropy, which were prepared under deposition angle of 0° and 40° .

4. Pulse-width dependent field-free switching of Pt/Co/Pt layers

The pulse-width-dependent field-free switching of a $\theta_{\text{dep}} = 40^\circ$ device was investigated to elucidate the influence from Joule heating, where the write current pulses were generated using an arbitrary pulse generator, Model Agilent 81110A. **Fig. S4(a)** represents the reduction of critical switching current (I_c) as increasing the pulse width of write current (t_{pulse}) from 0.03 ms (30 μ s)

to 900 ms. The I_c as functions of $\ln(t_{\text{pulse}}/\tau_0)$ shown in **Fig. S4(b)** demonstrates a typical linear behavior of thermal activation model⁴, which is described by **Equation S1**.

$$I_c = I_{c0} \left[1 - \frac{1}{\Delta} \ln \left(\frac{t_{\text{pulse}}}{\tau_0} \right) \right], \quad (\text{S1})$$

where I_{c0} is the zero-thermal critical switching current; $\Delta = U/k_B T$ parameterizes the thermal stability to describe the energy barrier (U) between magnetically UP and DOWN state, which is normalized by the product of Boltzmann constant (k_B) and temperature (T); $\tau_0 = 1$ ns is the intrinsic attempt time.

The extracted I_{c0} and Δ respectively are given as 13.07 mA and 27.72. Additionally, the observed linear relationship implies the field-free switching in this pulse range ($t_{\text{pulse}} > 0.03$ ms) is still governed by thermally-activated mechanism rather than dynamical (ballistic) switching.

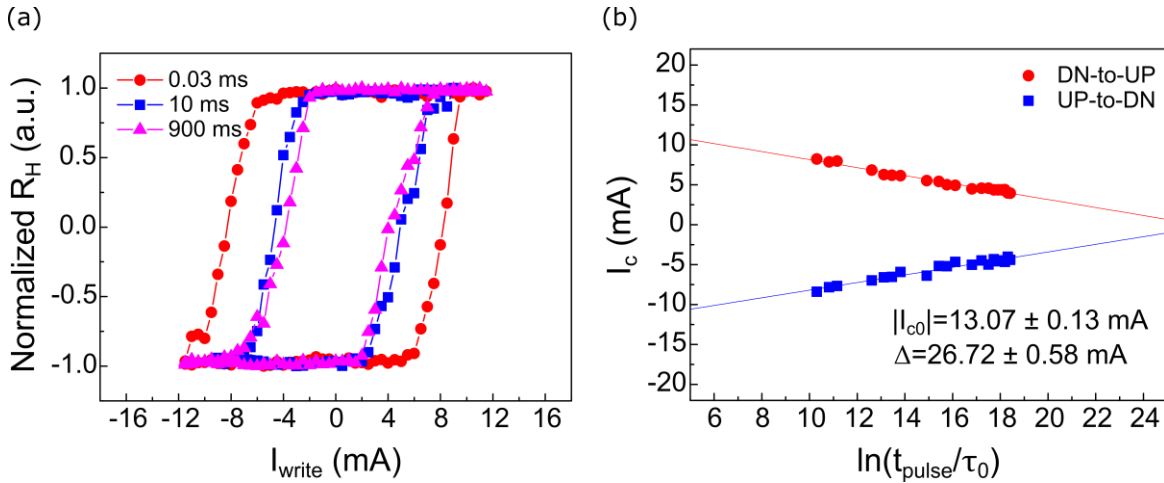


Fig. S4 Pulse-width dependent field-free switching of a device prepared by oblique angle deposition ($\theta_{\text{dep}} = 40^\circ$). **a** Representative switching loops by pulse width of 0.03 ms, 10 ms and 900 ms. **b** Linear relationship between $\ln(t_{\text{pulse}}/\tau_0)$ and critical switching current (I_c).

5. Pt/Co/Pt layers prepared by two-step fabrication

To identify which deposited layer plays a more decisive role to the resulting symmetry breaking and field-free switching, we prepared two different stacks by a two-step fabrication method, as illustrated in **Fig. S5**. For stack A (**Fig. S5(c)**), after preparing Ta(0.5)/Pt(5) layers in the same way as the devices of $\theta_{\text{dep}} = 40^\circ$ in the main text (using 40° holder and oblique angle deposition), we transferred the sample to the 0° -holder and deposited the rest of the layers, Co(0.6)/Pt(1)/Ta(2), by uniform deposition; for stack B (**Fig. S5(d)**), the Ta(0.5)/Pt(5) layers were uniformly deposited by using the 0° -holder, then transferred to the 40° -holder for the Co(0.6)/Pt(1)/Ta(2) depositions as the way of preparing the devices of $\theta_{\text{dep}} = 40^\circ$ in the main text (uniform deposition). The slightly thicker Co thickness used in these two stacks is because of the inevitable exposure to the atmosphere for Pt/Co interface when changing the holder, which changes the PMA condition in our experiments.

Compared to the devices in the main text shown in **Fig. S5(a,b)**, the field-free switching results shown in **Fig. S5(c,d)** show that only stack A can still achieve current-induced field-free switching, suggesting that the obliquely deposited Pt(5) serving as underlayer governs the symmetry breaking of this system. This also rules out the possible influence or introducing extra symmetry-breaking mechanism by preparing the Co layer on a rotating wedge-shape holder.

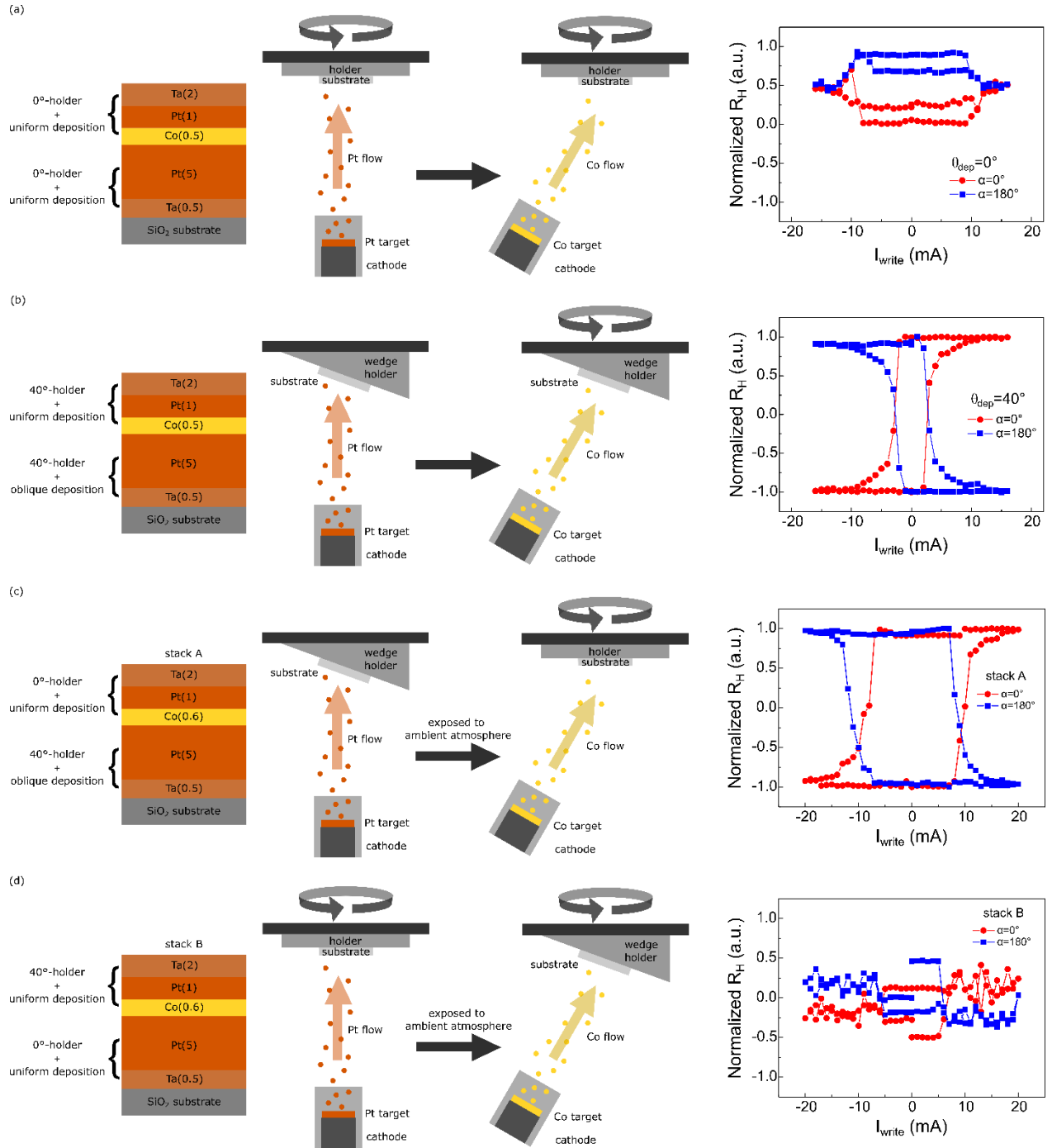


Fig. S5 The schematics of the fabrication details to prepare Pt/Co/Pt structures and the corresponding field-free switching results with $\alpha = 0^\circ, 180^\circ$ Hall bar arrangement. The uniform deposition indicates the sample is rotating during the deposition; otherwise, the oblique deposition is not. **a** the 0° -devices in the main text of Ta(0.5)/Pt(5)/Co(0.5)/Pt(1)/Ta(2) fully prepared by uniform deposition on the 0° -holder. **b** the 40° -devices in the main text of obliquely

deposited Ta(0.5)/Pt(5) and uniformly deposited Co(0.5)/Pt(1)/Ta(2) on the 40°-holder. **c** the stack A comprised of obliquely deposited Ta(0.5)/Pt(5) on the 40°-holder and uniformly deposited Co(0.6)/Pt(1)/Ta(2) on the 0°-holder. **d** the stack B of Ta(0.5)/Pt(5)/Co(0.6)/Pt(1)/Ta(2) fully prepared by uniform deposition on the 0°-holder.

6. Field-free switching uniformity of obliquely deposited Pt/Co/Pt layers

The device performance uniformity is verified through conducting the field-free switching measurements of two $\theta_{\text{dep}} = 40^\circ$ devices, which were positioned approximately 5 mm apart. The results depicted in **Fig. S6** suggest that both devices are capable of field-free switching and exhibit similar values for the switching current. Consequently, concerns regarding uniformity should be of minor significance that at this scale.

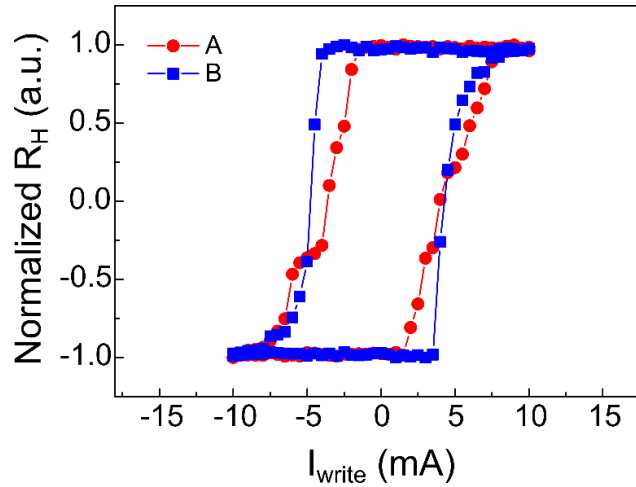


Fig S6. Switching uniformity check. Current-induced field-free switching loops of two devices prepared by $\theta_{\text{dep}} = 40^\circ$, which were positioned approximately 5 mm apart during the deposition process.

7. Switching field of spin-transfer-torque-like feature in the obliquely-deposited device

Under the zero in-plane magnetic field (H_{IP}) condition, the switching field (H_{sw}) under different applied currents ($I_{applied}$) in the AHE hysteresis loop measurement is defined as when the normalized Hall resistance changes its sign, and the loop center is defined as $\Delta H_{sw} \equiv \frac{1}{2}(H_{sw}^{DN-to-UP} + H_{sw}^{UP-to-DN})$. **Fig. S7** summarizes the switching behavior and shows a clear difference: a typical H_{sw} reduction from only the Joule heating for $\theta_{dep} = 0^\circ$ and a damping-like spin torque switching feature for $\theta_{dep} = 40^\circ$. Note that there is no obvious shift when the applied current $I_{applied}$ is less than ~ 2 mA.

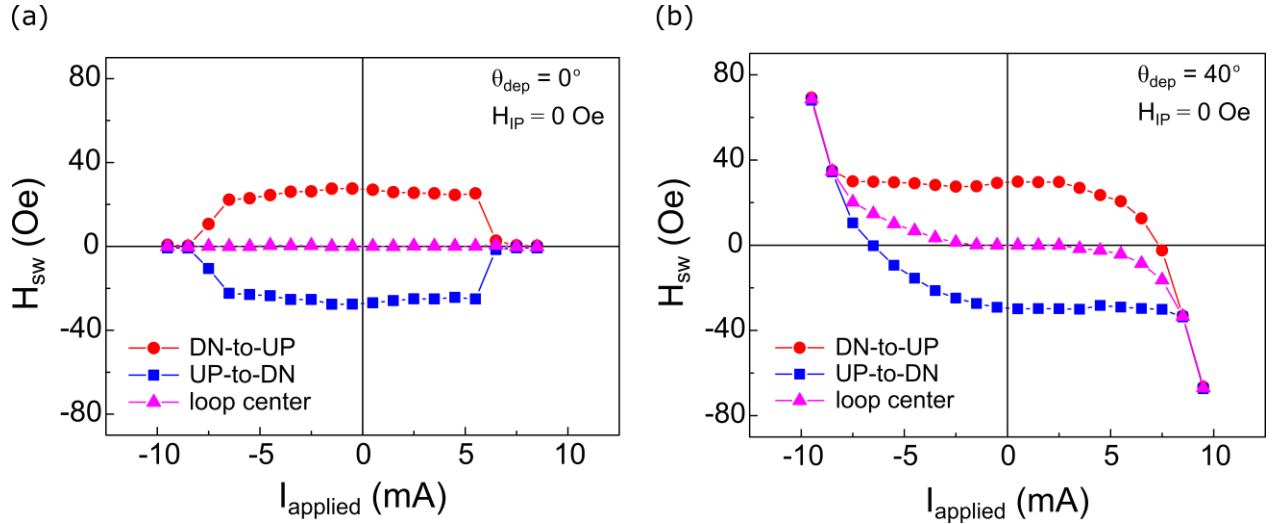


Fig. S7 The comparison of the switching field (H_{sw}) and loop center (ΔH_{sw}) under different applied current. a, b H_{sw} and ΔH_{sw} as functions of $I_{applied}$ for the devices prepared by $\theta_{dep} = 0^\circ$ and 40° . The measurements are characterized in an in-plane field-free condition ($H_{IP} = 0$ Oe).

Reference

- S1. C. F. Pai, M. Mann, A. J. Tan and G. S. D. Beach. Determination of spin torque efficiencies in heterostructures with perpendicular magnetic anisotropy. *Phys. Rev. B* **93**, 144409 (2016).
- S2. C. Y. Hu and C. F. Pai. Benchmarking of Spin–Orbit Torque Switching Efficiency in Pt Alloys. *Advanced Quantum Technologies*, 2000024 (2020).
- S3. S. J. Yun, E. S. Park, K. J. Lee and S. H. Lim. Accurate analysis of harmonic Hall voltage measurement for spin-orbit torques. *Npg Asia Materials* **9**, e449 (2017).
- S4. R. H. Koch, J. A. Katine and J. Z. Sun. Time-resolved reversal of spin-transfer switching in a nanomagnet. *Phys. Rev. Lett.* **92**, 088302 (2004).