

Supplementary Information

Two-dimensional Electronic Transport and Surface Electron Accumulation in MoS₂

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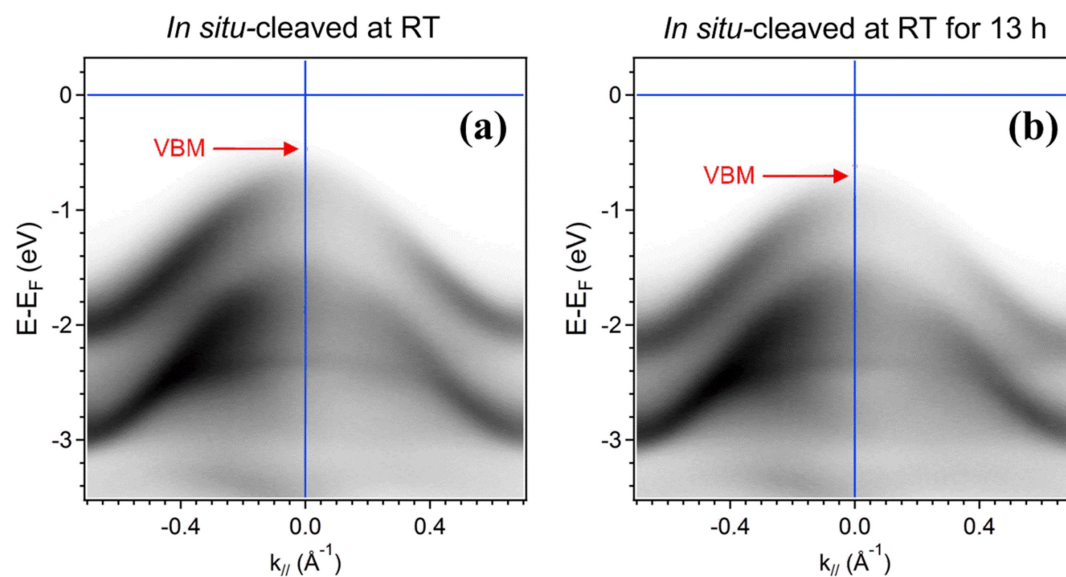
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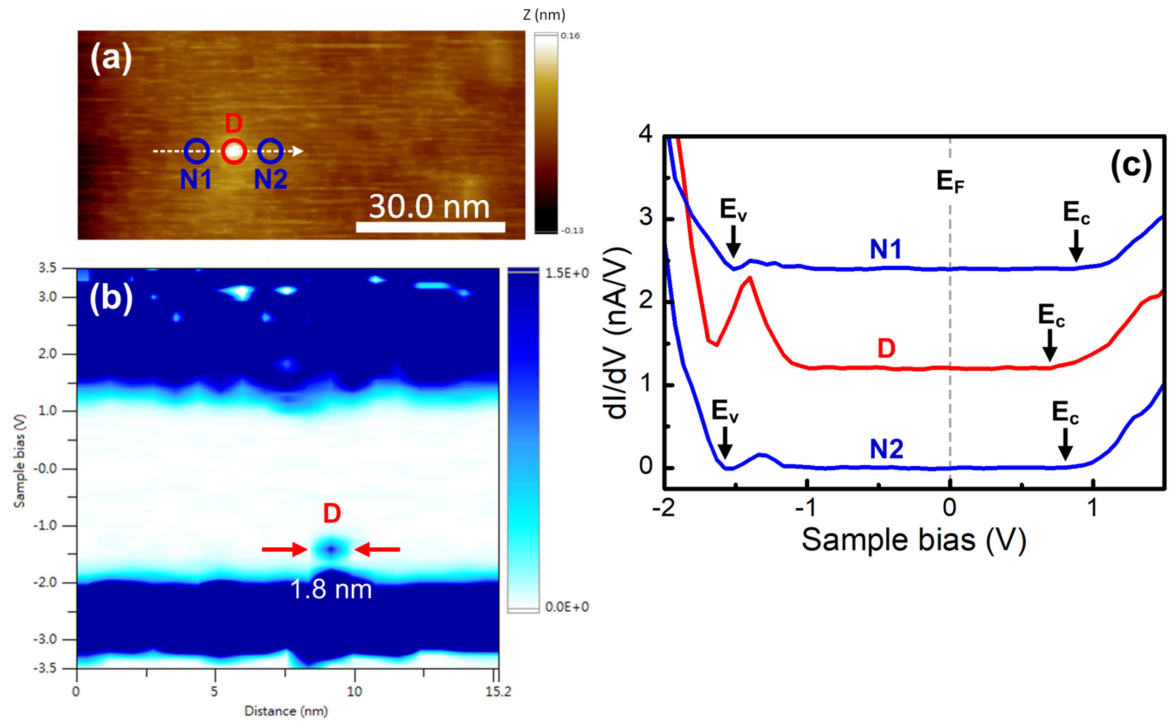
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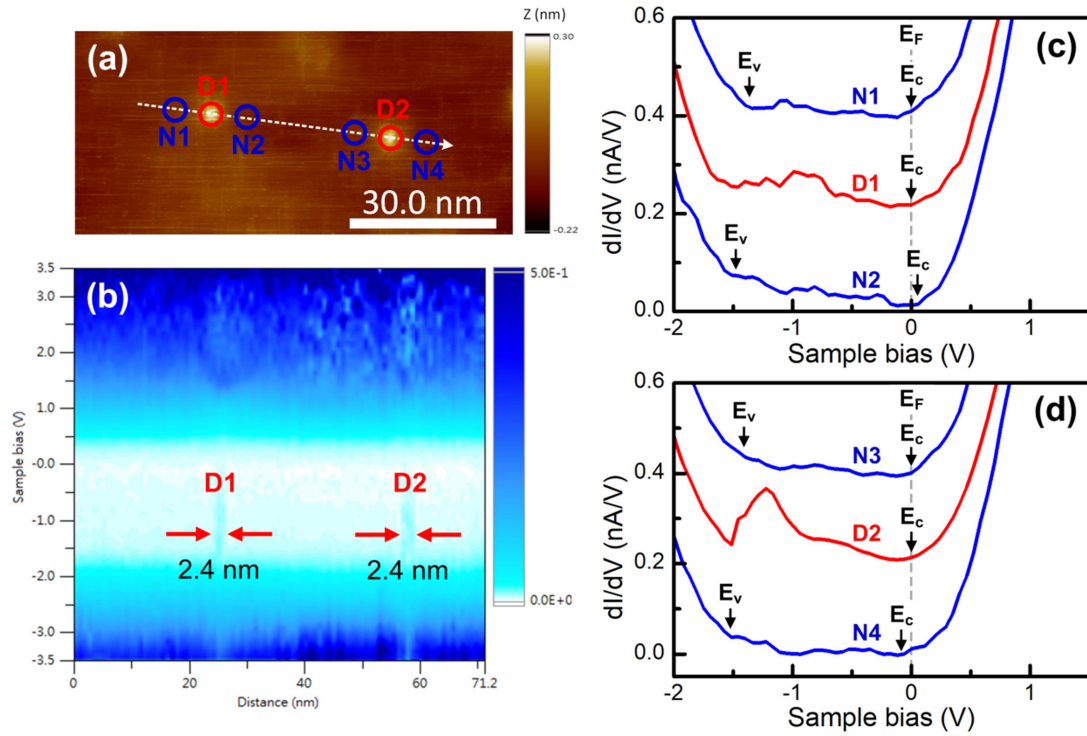


Supplementary Figure 1. Aging effect at room temperature (RT) of the *in situ*-cleaved fresh surfaces of a MoS₂ single crystal. The E versus $k_{\parallel}$ valance band measurements for the (a) *in situ*-cleaved surface at RT and (b) the *in situ*-cleaved surface at RT for 13 h of a MoS₂ crystal recorded with a 42-eV photon energy at RT. The observed overlap band structure can be attributed to a cleaved surface with two different crystal facets illuminated simultaneously by larger incident beam spot.



Supplementary Figure 2. STM/STS characterization of sulfur vacancies (SV) in the fresh MoS₂ surface. (a) STM image and (b) its STS mapping targeting an area with a point defect in the fresh MoS₂ surface. The white dash arrow in (a) indicates the scanning region of STS mapping. The red arrows in (b) indicate the influence region of the point defect labelled with D. (c) STS spectra measured at the defect center (position D) and its nearby area (positions N1 and N2) marked in (a).

Supplementary Note 1. By taking the STS spectra at the defect center (position D) and its nearby area (positions N1 and N2), we can clearly observe an extra peak close to the valance band (VB) edge centered at -1.4 V for the spectrum measured at position D as shown in **Supplementary Figure 2(c)**. The similar feature is not so significant at the positions near the defect site (N1 and N2). According to the literatures, the presence of the SV introduces the defect states in the bandgap close to the VB edge.¹⁻⁴ The Fermi level at the defect center (D) exhibiting the blue-shift for 0.12–0.20 eV compared to those of the nearby positions (N1 and N2) indicates the *n*-type nature of the SV, which is also consistent with the previous reports.⁵



Supplementary Figure 3. STM/STS characterization of sulfur vacancies (SV) in the non-fresh MoS₂ surface. (a) STM image and (b) its STS mapping targeting an area with two point defects in the pristine (non-fresh) MoS₂ surface. The white dash arrow in (a) indicates the scanning region of STS mapping. The red arrows in (b) indicate the influence regions of the point defects labelled with D1 and D2. (c) STS spectra measured at the defect center (position D1) and its nearby area (positions N1 and N2) marked in (a). (d) STS spectra measured at the defect center (position D2) and its nearby area (positions N3 and N4) marked in (a).

Supplementary Note 2. According to the STS spectra in [Supplementary Figures 3\(c\) and 3\(d\)](#), we can also observe the similar features close to the valence band edge at the positions of defect centers (D1 and D2). The intensity of the extra peak in the non-fresh surface is not as high as that in the fresh sample, which is probably due to the long-term exposure in air. Santosh et al suggest that the oxygen adsorption could suppress the states originating from the SV.⁴ The energy distribution and its density of states (DOS) could be slightly changed by the interaction of SV and oxygen molecules. The increase of the SV density could make a broader distribution of the defect states.^[2] These statements probably can explain that the SV-related features are

slightly different in energy position and relative intensity in the STS spectra shown in [Supplementary Figures S2\(c\), S3\(c\) and S3\(d\)](#).

Supplementary Note 3. Calculation of surface electron diffusion length in MoS₂.

The surface electron accumulation originates from the electron injection from the surface (donor-like surface states) into the bulk. The surface-bulk interface is somewhat similar to the n^{++} - n junction. To understand to what thickness the electron accumulation persists, we can estimate the diffusion length of electron (L_n) in the intrinsically n -type MoS₂. According to the one-dimensional continuity equation at steady state: $D_n \frac{d^2(\delta n)}{dx^2} - \frac{\delta n}{\tau_n}$, the excess electron concentration (δn) as a function of space (x) is written as $\delta n(x) = \delta n(0) \exp(-x/L_n)$, where $\delta n(0)$ is the excess electron concentration at $x = 0$ and $L_n = (D_n \tau_n)^{1/2}$, where D_n is the diffusion coefficient of electron and τ_n (~ 1 ns)⁶ is the electron lifetime.⁷ The D_n of MoS₂ can be calculated by the Einstein equation, $D_n/\mu_n = kT/q$, where μ_n is the electron mobility, k is Boltzmann's constant, T is the temperature set at 300 K, and q is the elementary charge.

The layer material like MoS₂ has strong anisotropic transport properties. The conductivity and effective mass of electron perpendicular to the c -axis (in-plane) is approximately three orders of magnitude higher than that along c -axis (out-of-plane).⁸ According to the references, the in-plane mobility values of MoS₂ bulk^{9,10} are 32–100 cm²V⁻¹s⁻¹ and so the out-of-plane mobility values are inferred at the range of 0.032–0.1 cm²V⁻¹s⁻¹. Because the electron injection is along the c -axis, the out-of-plane mobilities were adopted for the D_n calculation. The estimated D_n are 8.3×10^{-4} – 2.6×10^{-3} cm²s⁻¹ and the obtained L_n locates in the range of 9.1–16 nm.

Supplementary References

1. Fuhr, J. D., Saul, A., Sofo, J. O. Scanning tunneling microscopy chemical signature of point defects on the MoS₂(0001) surface. *Phys. Rev. Lett.* **92**, 026802 (4 pages) (2004).
2. Qiu, H. *et al.* Hopping transport through defect-induced localized states in molybdenum disulphide. *Nature Commun.* **4**, 2642 (6 pages) (2013).
3. Santosh, K. C., Longo, R. C., Addou, R., Wallace, R. M., Cho, K. Impact of intrinsic atomic defects on the electronic structure of MoS₂ monolayers. *Nanotechnology* **25**, 375703 (6 pages) (2014).
4. Akdim, B., Pachter, R., Mou S. Theoretical analysis of the combined effects of sulfur vacancies and analyte adsorption on the electronic properties of single-layer MoS₂. *Nanotechnology* **27**, 185701 (10 pages) (2016).
5. McDonnell, S., Addou, R., Buie, C., Wallace, R. M., Hinkle, C. L. Defect-dominated doping and contact resistance in MoS₂. *ACS Nano* **8**, 2880-2888 (2014).
6. Wang, H., Zhang, C., Rana, F. Surface Recombination Limited Lifetimes of Photoexcited Carriers in Few-Layer Transition Metal Dichalcogenide MoS₂. *Nano Lett.* **15**, 8204-8210 (2015).
7. Neamen, D. A. *Semiconductor Physics and Devices: Basic Principles*, 3rd edition (McGraw-Hill Inc., New York, **2003**), Chap. 6, pp. 206-207.
8. Guha Thakurta, S. R., Dutta, A. K. Electrical conductivity, thermoelectric power and Hall effect in *p*-type molybdenite (MoS₂) crystal. *J. Phys. Chem. Solids* **44**, 407-416 (1983).
9. Tiong, K. K., Liao, P. C., Ho, C. H., Huang, Y. S. Growth and characterization of rhenium-doped MoS₂ single crystals. *J. Crystal Growth* **205**, 543-547 (1999).
10. Fivaz, R., Mooser, E. Mobility of charge carriers in semiconducting layer structures. *Phys. Rev.* **163**, 743-755 (1967).