

Supplementary Information for

General strategy for activating ferroelectricity in bilayer 2D materials with intercalating inert atoms

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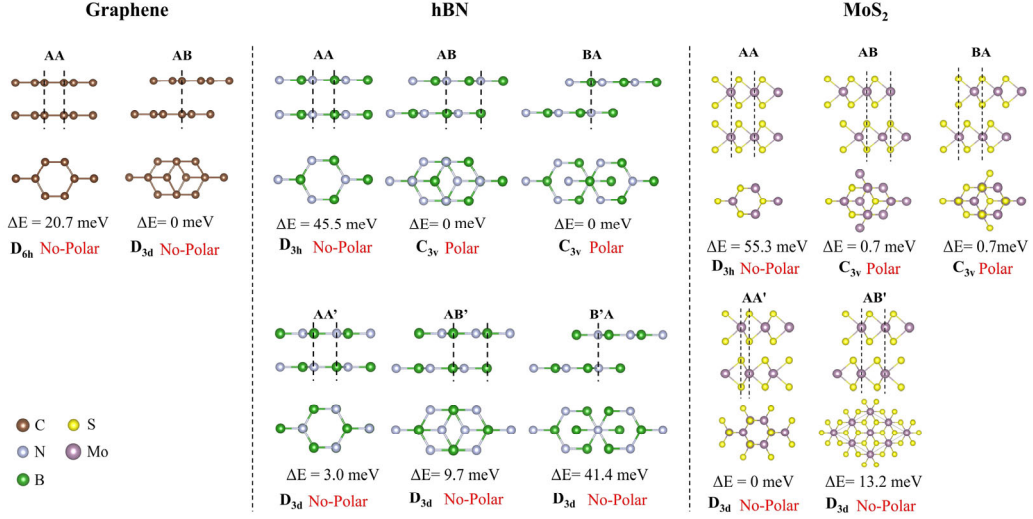


Fig. S1. Side and top views of atomic structures for bilayer graphene, hBN, and MoS₂ in different stacking configurations, along with their relative energies (ΔE), point group symmetries, and polarities.

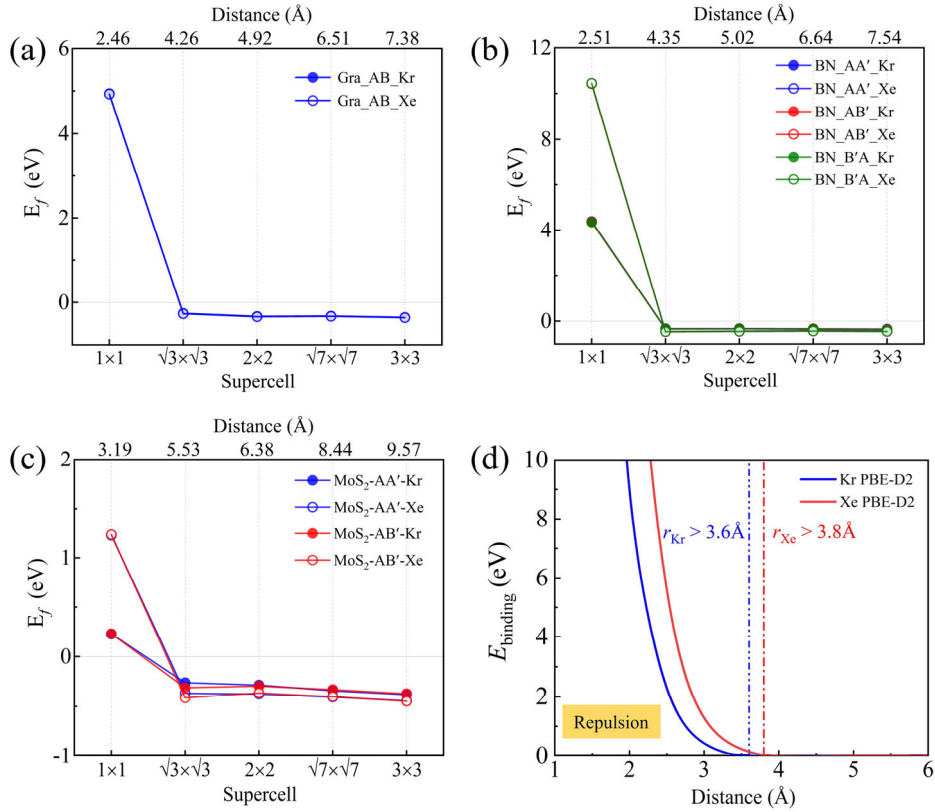


Fig. S2. Formation energy (E_f) of inert atoms intercalated in bilayer (a) graphene, (b) h-BN, and (c) MoS₂ as a function of supercell size and inert-atom distance. **d** Binding energy (E_{binding}) between two inert atoms as a function of supercell size and inert-atom distance. The vertical dashed line marks the critical distances where $E_{\text{binding}} = 0$. The binding energy approaches zero at large separations ($> 4 \text{ Å}$), indicating negligible interaction, while its increase at shorter distances reflects the emergence of strong repulsion.

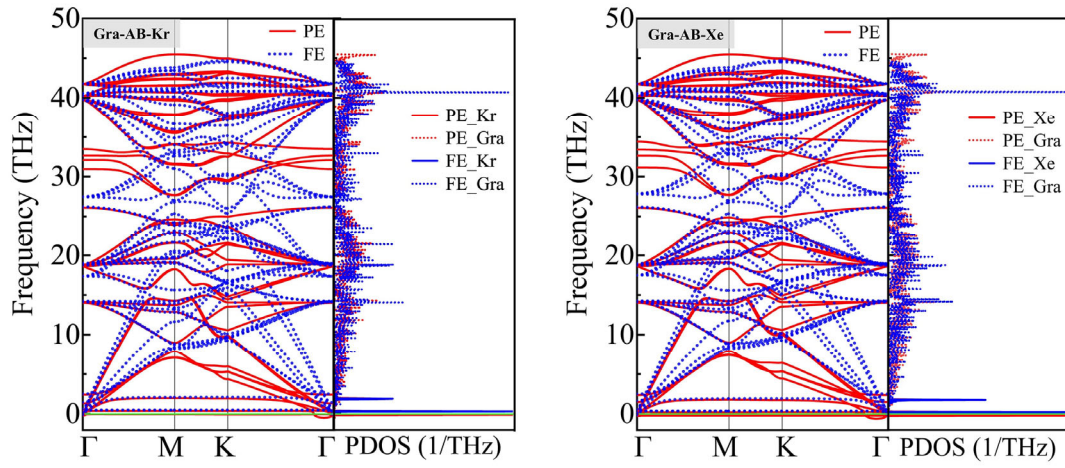


Fig. S3. Phonon spectra and projected DOS of FE and PE phases for Kr/Xe-intercalated graphene bilayers. The green horizontal line represents the phonon mode with zero energy.

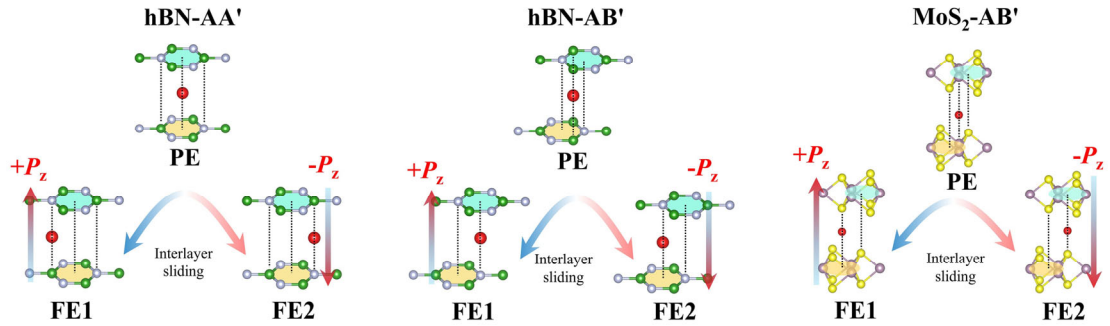


Fig. S4. Side views of the FE and PE states of inert-atom-intercalated AA'- and AB'-stacked bilayer hBN, AB'-stacked bilayer MoS₂. The red balls represent inert atoms Kr/Xe.

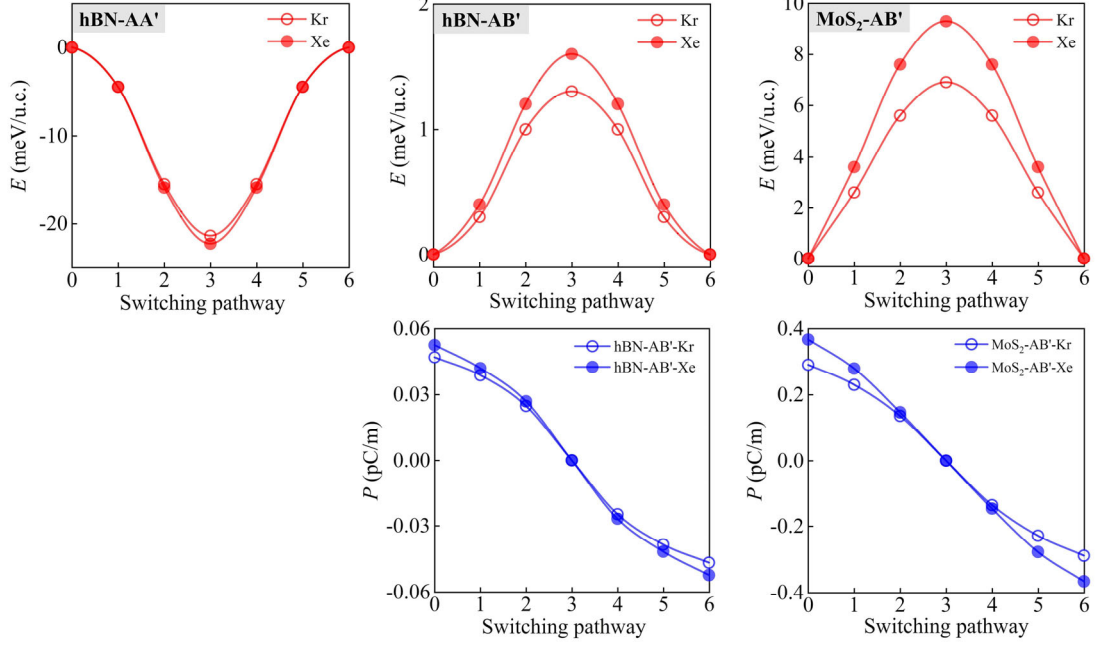


Fig. S5. Energy barriers and OOP electric polarization of inert-atom-intercalated AA'- and AB'-stacked bilayer hBN, AB'-stacked bilayer MoS₂. Due to the instability of the FE state in Xe-intercalated AA'-stacked hBN, the polarization associated with FE switching is absent.

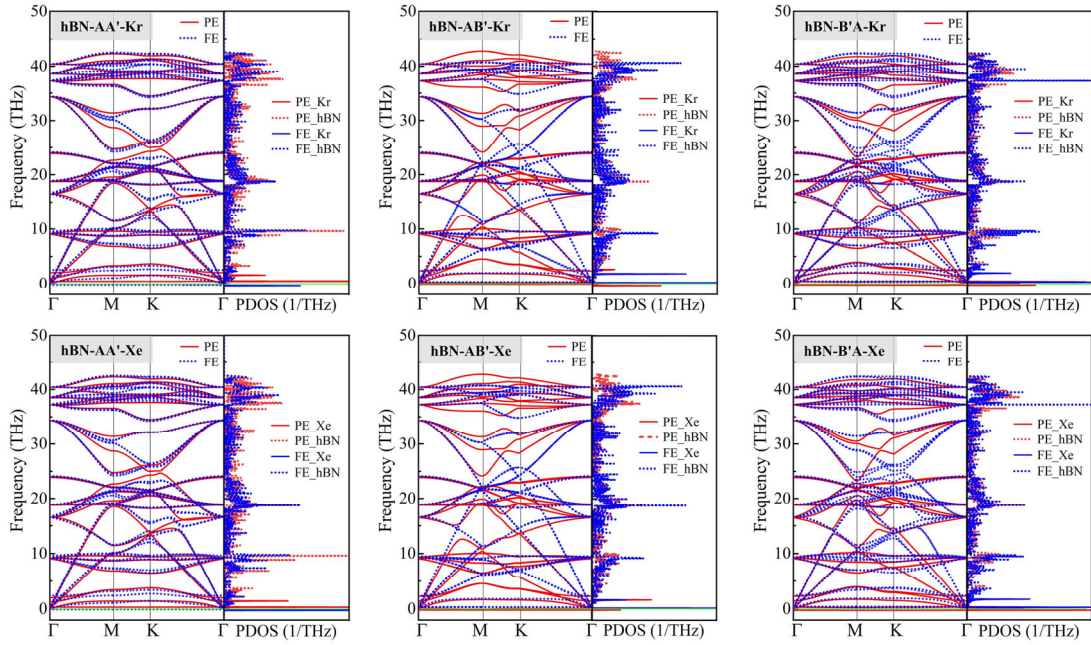


Fig. S6. Phonon spectra and projected DOS of FE and PE phases for Kr/Xe-intercalated hBN bilayers. The green horizontal line represents the phonon mode with zero energy.

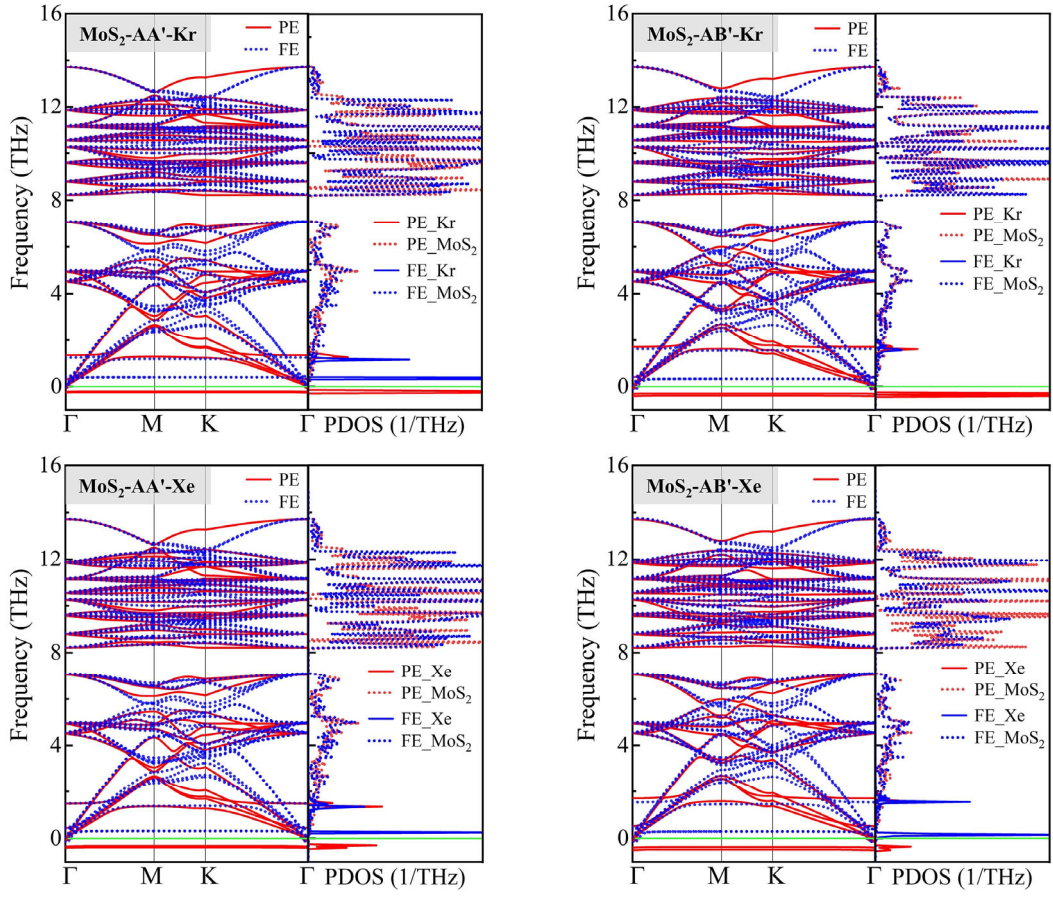


Fig. S7. Phonon spectra and projected DOS of FE and PE phases for Kr/Xe-intercalated MoS₂ bilayers. The green horizontal line represents the phonon mode with zero energy.

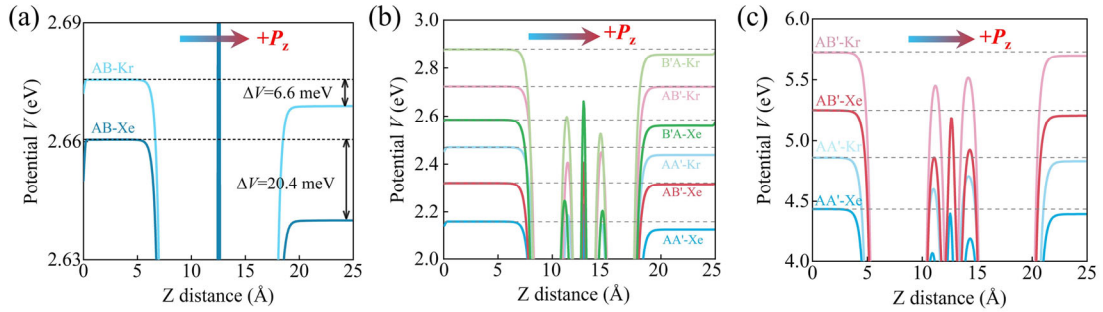


Fig. S8. Planar-averaged electrostatic potential of the FE states for inert-atom-intercalated bilayers.

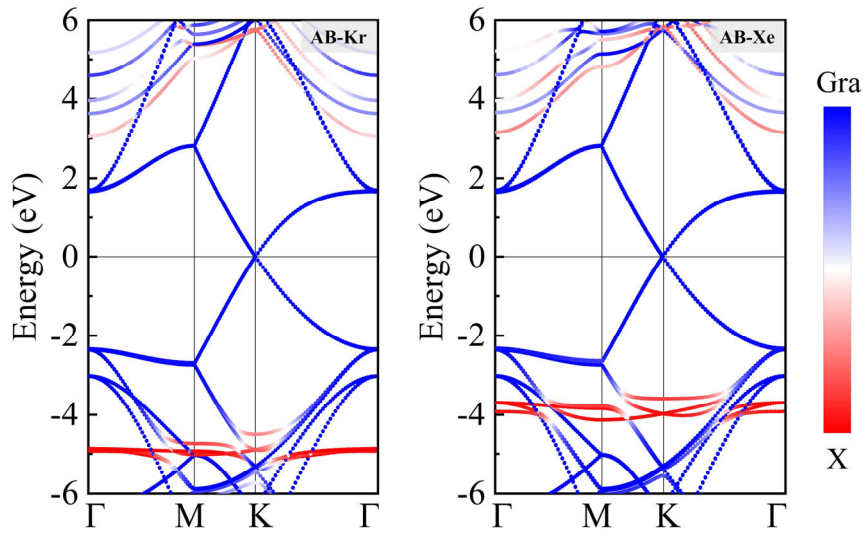


Fig. S9. Atomically resolved band structures for inert-atom-intercalated bilayer graphene.

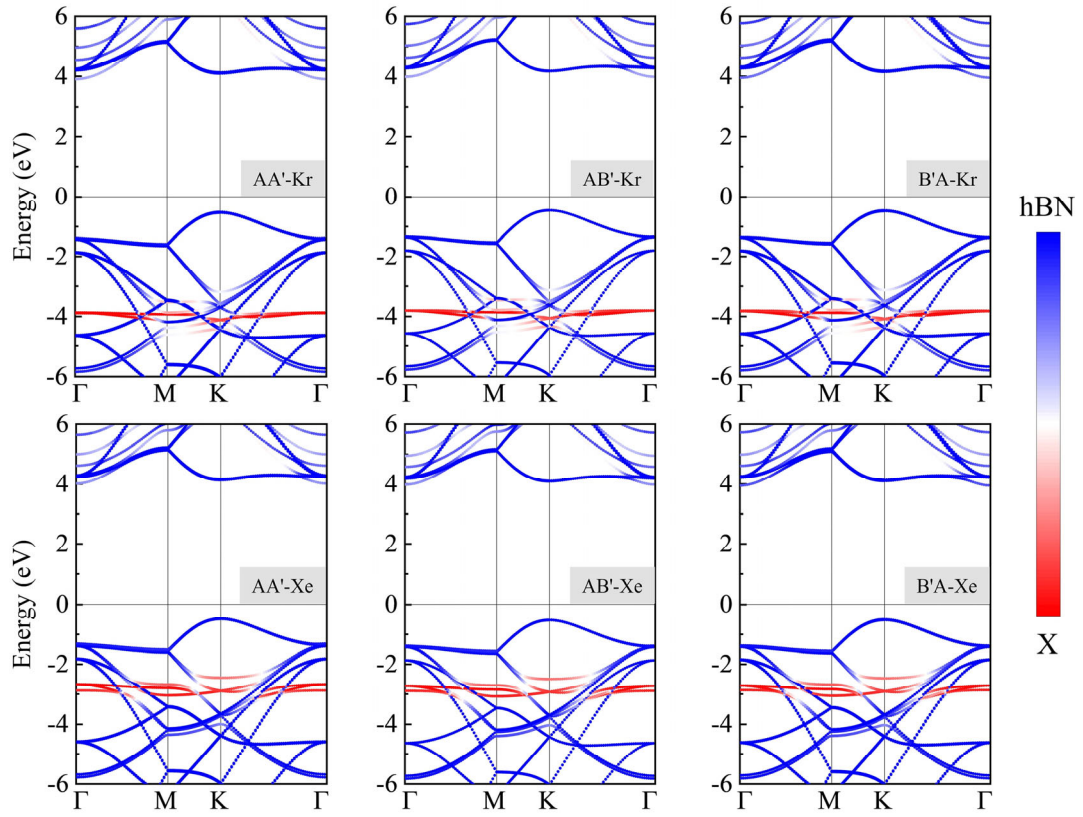


Fig. S10. Atomically resolved band structures for inert-atom-intercalated bilayer hBN.

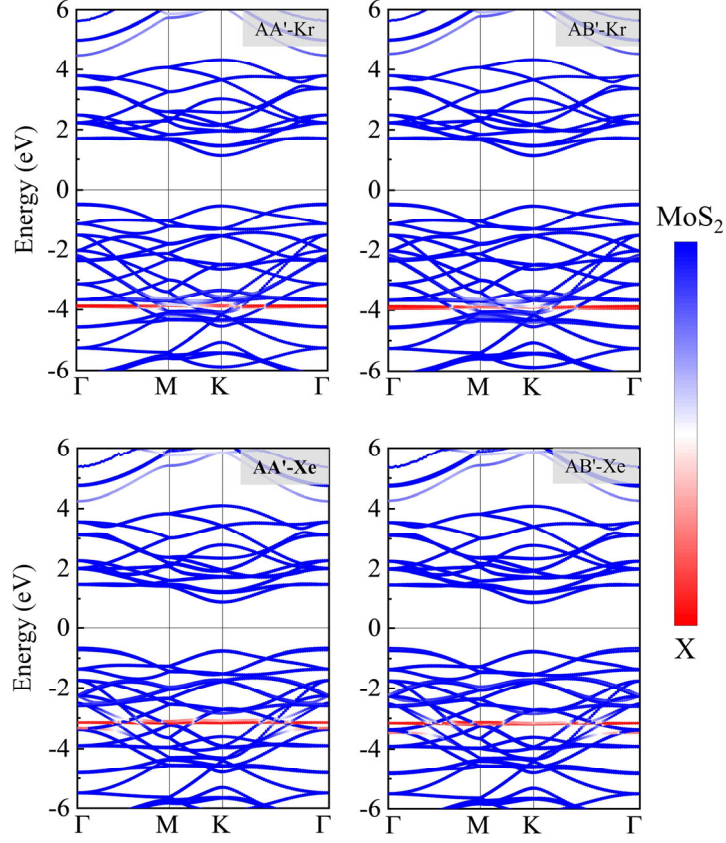


Fig. S11. Atomically resolved band structures for inert-atom-intercalated bilayer MoS₂.

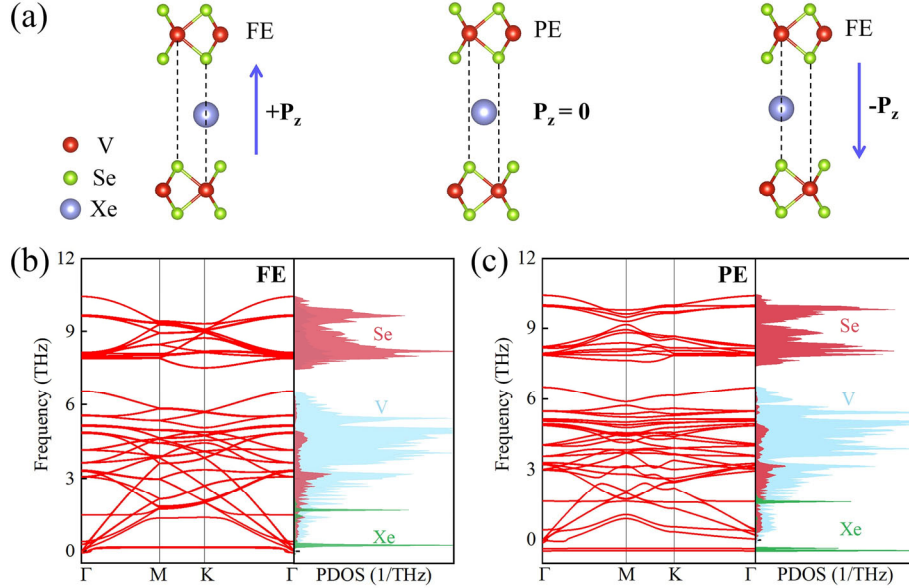


Fig. S12. (a) Side views of the FE and PE states of Xe-intercalated AA'-stacked 2×2×1 bilayer VSe₂. The FE polarization P_z is 0.25 pC/m. Phonon spectra and projected DOS of (b) FE and (c) PE states. No imaginary phonon modes are observed in the FE phase, whereas the PE phase exhibits clear imaginary branches.

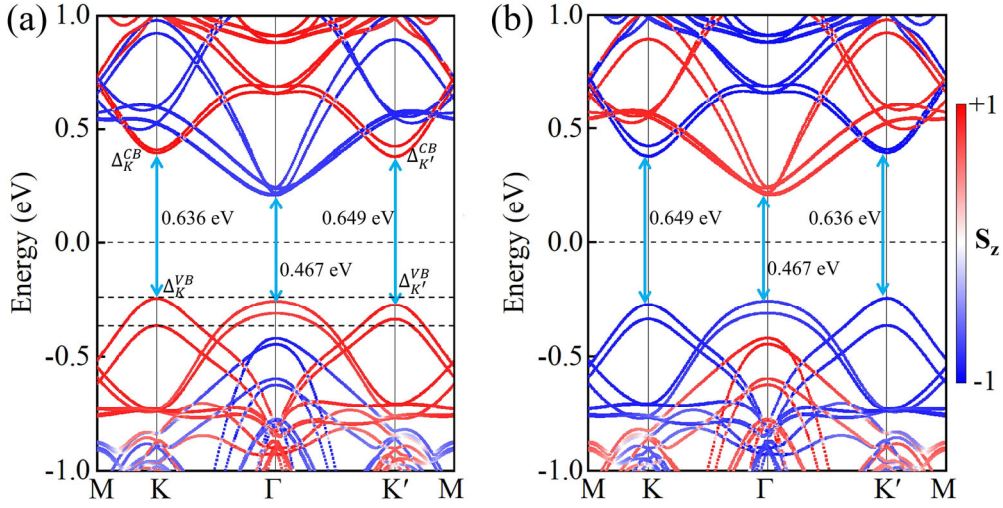


Fig. S13. Spin-resolved band structures of Xe-intercalated bilayer VSe₂ with spin-orbit coupling under (a) +P_z and (b) -P_z states. The color bar represents the spin contribution of the energy band, in which +1 and -1 represent two opposite spin directions (S_z), respectively. The PBE+ U ($U = 1.0$ eV) method was adopted. The Fermi level is set to 0 eV. $\Delta_K^{CB} = 13$ meV, $\Delta_{K'}^{CB} = 45$ meV, $\Delta_K^{VB} = 122$ meV, and $\Delta_{K'}^{VB} = 65$ meV.

Linear relationship between OOP polarization and electrostatic potential difference

To elucidate the origin of out-of-plane ferroelectricity induced by intercalated inert atoms in bilayers, we establish the relation between the spontaneous polarization P and the electrostatic potential difference ΔV . Upon intercalation of noble gas atoms (e.g., Kr, Xe) into the van der Waals gap, the system symmetry is reduced from centrosymmetric D_{3d} to polar C_{3v} , leading to asymmetric charge redistribution and net electric polarization.

In a simplified dipole model, this asymmetric charge distribution can be treated as an electric dipole formed by partial charge displacement $\pm q$ across an effective separation distance d , corresponding to the vertical shift between the top and bottom layers or charge centers. The total dipole moment per unit cell is given by $p = q \cdot d$, and the spontaneous polarization is defined as:

$$P = p/A = (q \cdot d)/A, \quad (S1)$$

where A is the in-plane area of the unit cell. The induced polarization generates an internal electric field $E = P/(\epsilon_0 \epsilon_r)$, which in turn gives rise to an electrostatic potential difference:

$$\Delta V = E \cdot d = \frac{P \cdot d}{\epsilon_0 \epsilon_r} \Rightarrow P = \frac{\epsilon_0 \epsilon_r}{d} \cdot \Delta V, \quad (S2)$$

where ϵ_0 and ϵ_r are the vacuum dielectric constant and effective dielectric constant, respectively. The proportionality factor $\alpha = \epsilon_0 \epsilon_r / d$ depends on the interlayer distance d and the effective dielectric constant ϵ_r . Table S1 lists the interlayer spacing d and dielectric constants ϵ_r of graphene, hBN, and MoS₂. Substituting these parameters into the proportionality factor reveals that the systems remain within the linear polarization response regime. Although the precise values of ϵ_r and d may vary slightly between different materials, the linear trend persists due to the self-consistent nature of the polarization and electrostatic potential.

Table S1. Interlayer spacing and dielectric constants of graphene, hBN, and MoS₂ reported in literatures.

Materials	Distance d (Å)	Effective dielectric constant ϵ_r	Ref.
Graphene	3.4	3	[1,2]
hBN	3.5 ~ 3.8	3.3	[3,4]
MoS ₂	3.1 ~ 3.7	3.4	[5,6]

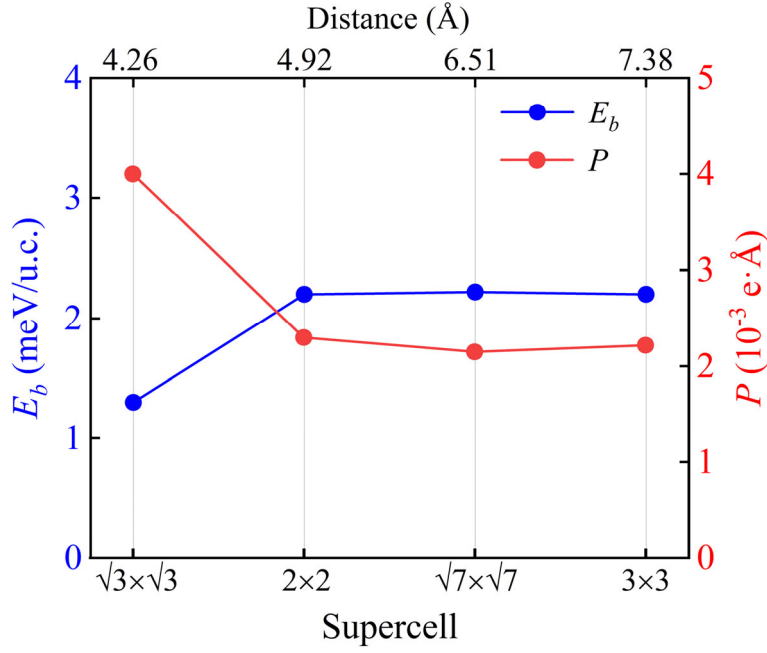


Fig. S14. Switching barriers (E_b) and FE polarizations (P) of Xe-intercalated AB-stacked bilayer graphene as a function of supercell size and inert-atom distance.

References:

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