

Obtaining $V_2(PO_4)_3$ by sodium extraction from single-phase $Na_xV_2(PO_4)_3$ ($1 < x < 3$) positive electrode materials

In the format provided by the authors and unedited

Reaching $V_2(PO_4)_3$ by Sodium Extraction from Single-Phase $Na_xV_2(PO_4)_3$ ($1 < x < 3$) Positive Electrode Materials

Supplementary Information

Table of Contents

Figures S1 : Temperature-controlled *in situ* XRD patterns of various mixtures of $Na_3V_2(PO_4)_3$ and $Na_1V_2(PO_4)_3$

Figure S2 : SEM images

Figure S3 : Thermogravimetric Analysis

Figure S4 : XRD patterns measured under different conditions

Figure S5 : Predicted phase diagram of $Na_xV_2(PO_4)_3$

Figure S6 : Computed DOS of different phases $Na_1V_2(PO_4)_3$

Figure S7 : Computed DOS of different phases $Na_1V_2(PO_4)_3$

Figure S8 : Structural illustration of Na distribution in $Na_xV_2(PO_4)_3$

Table S1 : Rietveld refinements of $Na_2V_2(PO_4)_3$

Source data tables for Figs. 1–5

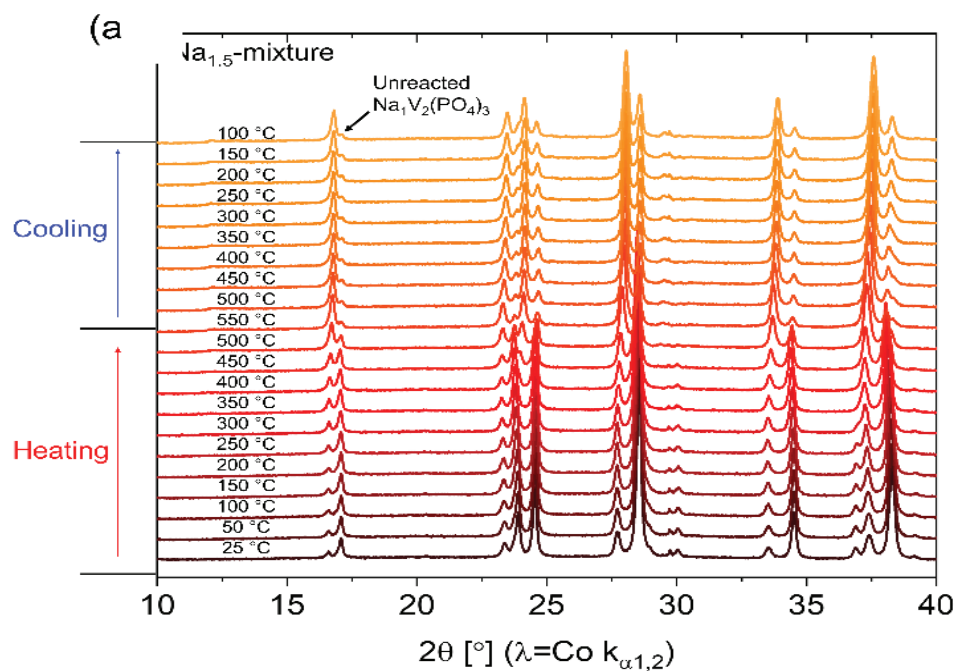


Figure S1a. Temperature-controlled *in situ* XRD patterns of the mixture of 0.25 Na₃V₂(PO₄)₃ and 0.75 Na₁V₂(PO₄)₃ collected every 50 °C upon heating up to 550 °C then cooling down to 100 °C.

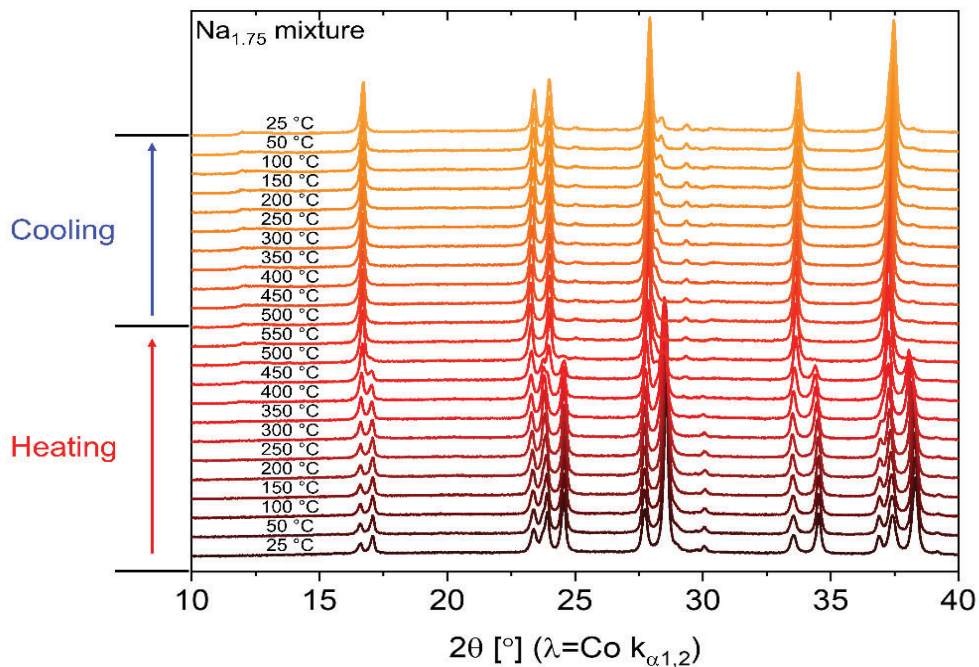


Figure S1b. Temperature-controlled *in situ* XRD patterns for the mixture of 0.375 Na₃V₂(PO₄)₃ and 0.625 Na₁V₂(PO₄)₃ collected every 50 °C upon heating up to 550 °C then cooling down to 25 °C.

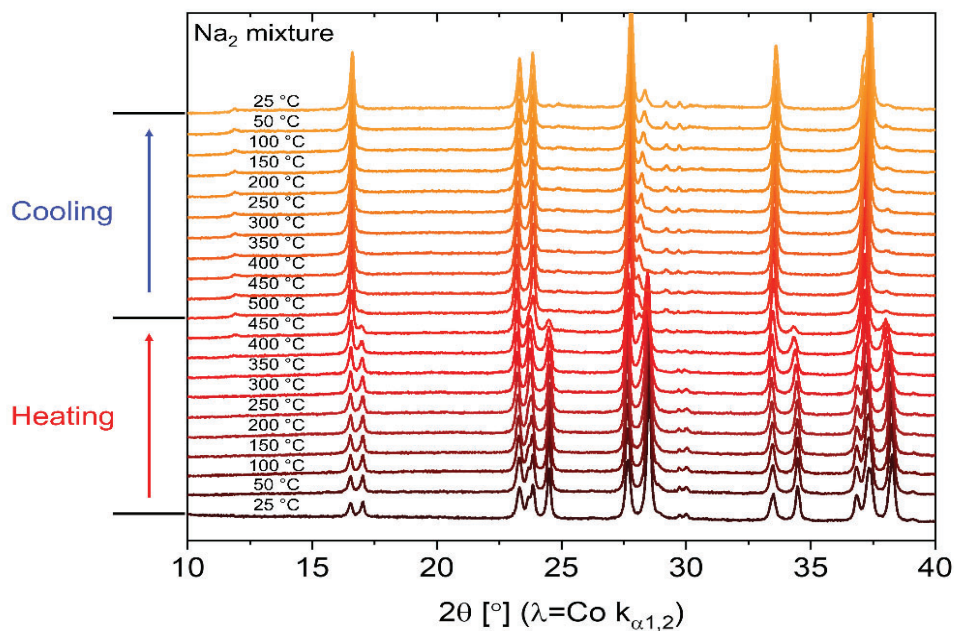


Figure S1c. Temperature-controlled *in situ* XRD patterns for the mixture of 0.5 $\text{Na}_3\text{V}_2(\text{PO}_4)_3$ and 0.5 $\text{Na}_1\text{V}_2(\text{PO}_4)_3$ collected every 50 °C upon heating up to 500 °C then cooling down to 25 °C.

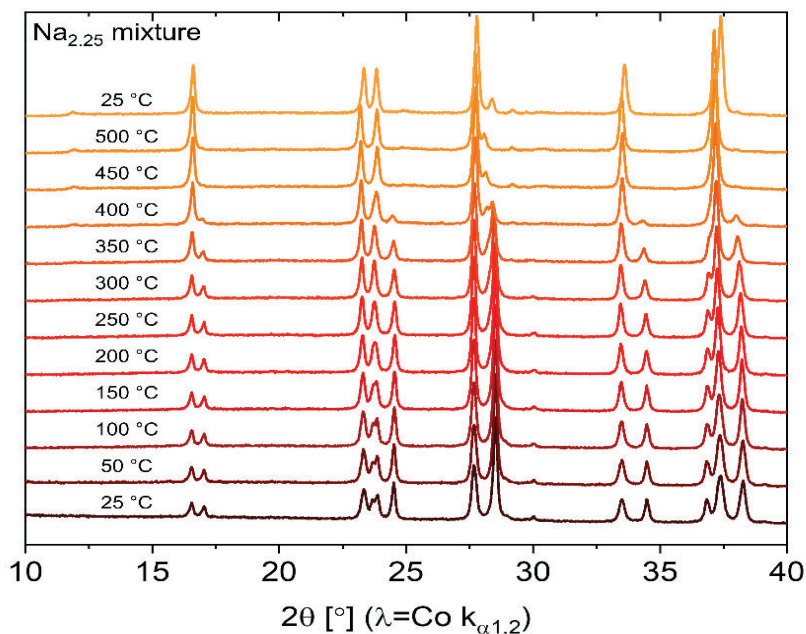


Figure S1d. Temperature-controlled *in situ* XRD patterns for the mixture of 0.625 $\text{Na}_3\text{V}_2(\text{PO}_4)_3$ and 0.375 $\text{Na}_1\text{V}_2(\text{PO}_4)_3$ collected every 50 °C upon heating up to 500 °C then measured at 25 °C after cool down.

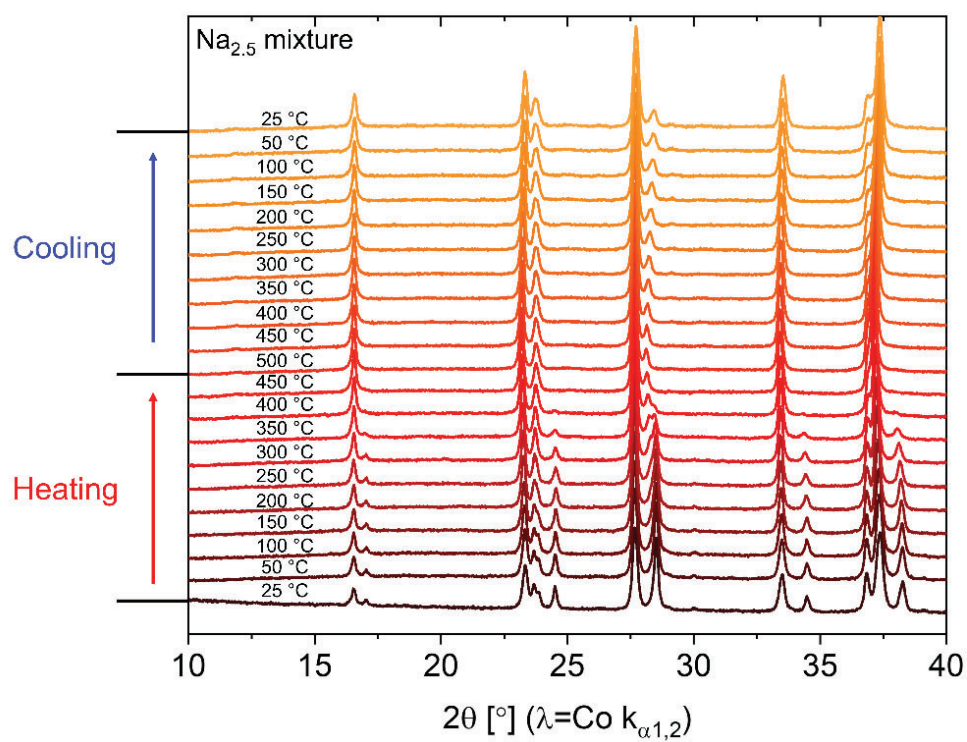


Figure S1e. Temperature-controlled *in situ* XRD patterns for the mixture of 0.75 $\text{Na}_3\text{V}_2(\text{PO}_4)_3$ and 0.25 $\text{Na}_1\text{V}_2(\text{PO}_4)_3$ collected every 50 °C upon heating up to 500 °C then cooling down to 25 °C.

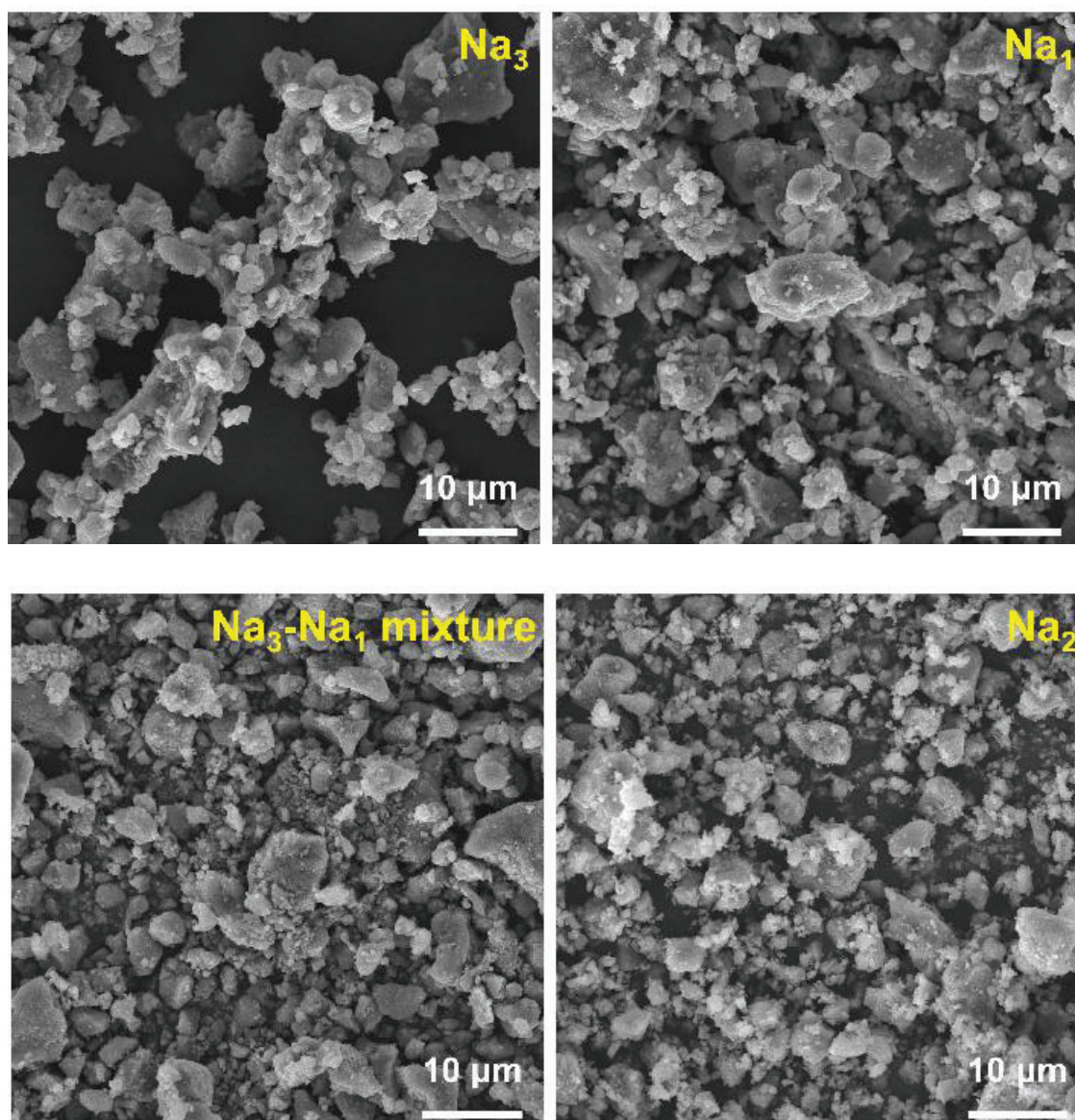


Figure S2. SEM images of $\text{Na}_3\text{V}_2(\text{PO}_4)_3$ (Na_3), $\text{Na}_1\text{V}_2(\text{PO}_4)_3$ (Na_1), mixture of $\text{Na}_3\text{V}_2(\text{PO}_4)_3$ and $\text{Na}_1\text{V}_2(\text{PO}_4)_3$ ($\text{Na}_3\text{-Na}_1$ 1), and $\text{Na}_2\text{V}_2(\text{PO}_4)_3$ after annealing at 500 °C (Na_2).

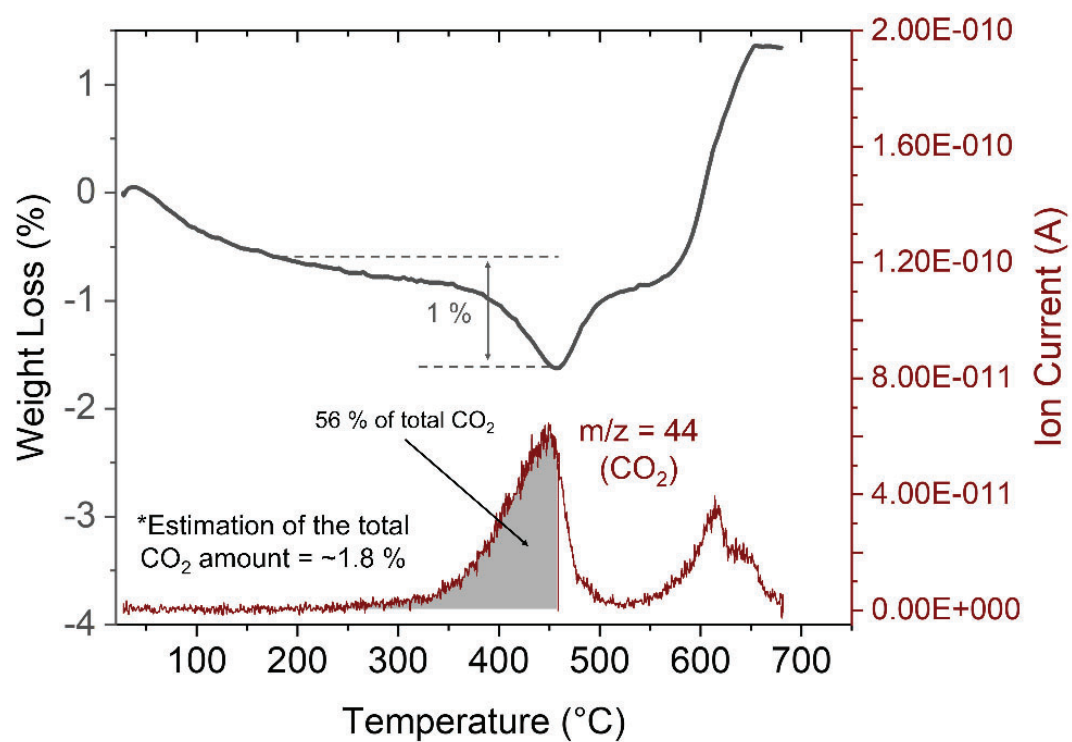
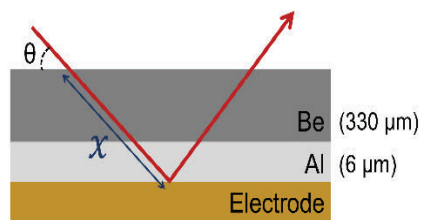
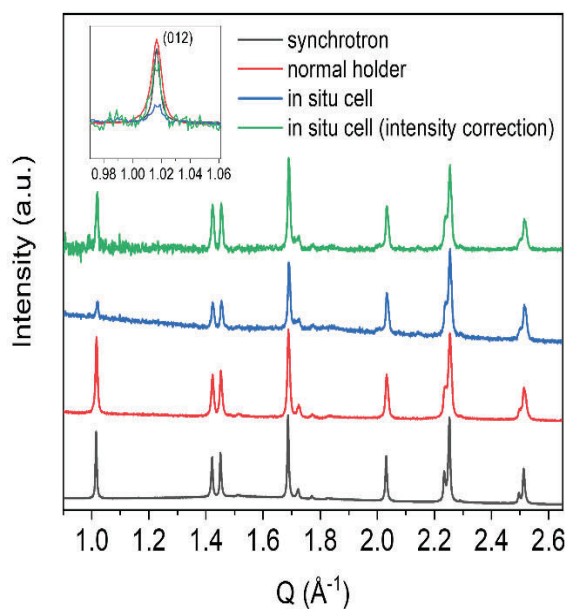


Figure S3. Thermogravimetric analysis, under air, combined with mass spectroscopy for the pristine $\text{c-Na}_3\text{V}_2(\text{PO}_4)_3$.



Corrected intensity was deduced from the mass attenuation coefficient based on the Cu K α radiation

$$I_o = I / \exp(-\mu \times 2x)$$

I = transmitted intensity

I_o = incident (corrected) intensity

μ = linear attenuation coefficient (cm⁻¹)

* $\mu_{Be} = 2.035$, and $\mu_{Al} = 133.92$

$2x$ = path length (cm)

Figure S4: Comparison of the XRD patterns of the chemically-prepared Na₂V₂(PO₄)₃ obtained within synchrotron X-ray, lab X-ray with normal holder, lab X-ray with an *in situ* cell, and lab X-ray with an *in situ* cell after the intensity correction (mass attenuation coefficient).

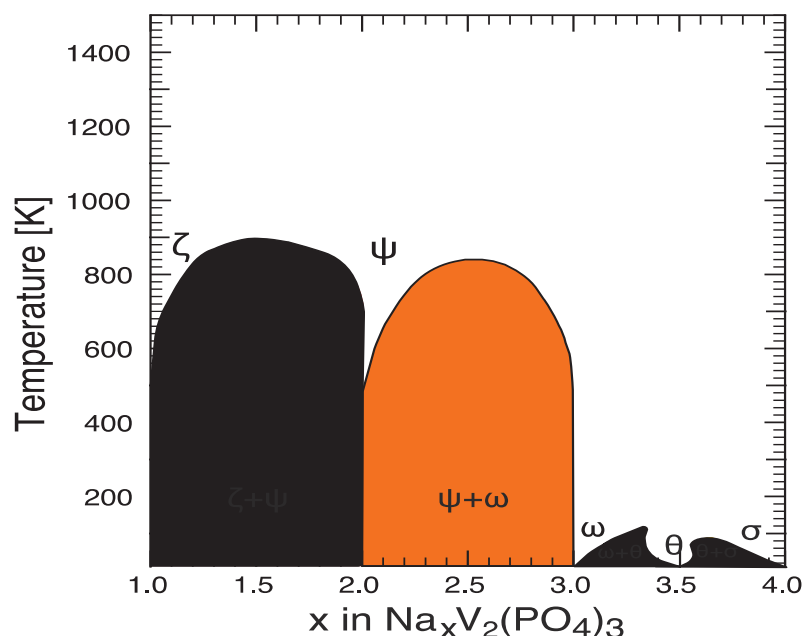


Figure S5. Predicted phase diagram of $\text{Na}_x\text{V}_2(\text{PO}_4)_3$ at variable Na contents (x) and temperatures (T). Solid lines show the phase boundaries. Precisely, ζ , ψ , and ω correspond to single-phase regions at Na compositions of $x = 1, 2$ and 3 , respectively. At 0 K , ζ , ψ , and ω exist as the stable phases and have the same $\text{Na}(1) / \text{Na}(2)$ occupations as the e-GS phases shown in **Table 2** of the main manuscript. Reproduced from Ref. [1] with permission from the Royal Society of Chemistry.¹

The phase diagram of **Figure S2**, above, justifies the stability region of all single phases. For example, Na_2VP phase is stable and thermodynamically preserved over wide temperature region (i.e., $0 - 800\text{ K}$). Additionally, the derived configurational entropy $S(x)$ in Ref. [1] also shows a local minimum at $x = 1, 2$, and 3 , respectively, which agrees with the appearance of single phases ζ , ψ , and ω in this work. Notably, $S(x)$ in Ref. 16 indicates all the derived single phases (i.e., ζ , ψ , and ω) are line compounds. All these results emphasize the formation of a single phase at RT. By the experimental procedure highlighted in this work (mixing and then annealing), we can further identify additional single phases, such as $\text{Na}_{1.5}\text{VP}$, $\text{Na}_{1.75}\text{VP}$, $\text{Na}_{2.25}\text{VP}$ and $\text{Na}_{2.5}\text{VP}$, which compensate for the formation of line compounds being stable at room temperature.

Density of States of $\text{Na}_1\text{V}_2(\text{PO}_4)_3$

Figure S6 and **Figure S7** show the computed density of states (DOS) for different phases of $\text{Na}_1\text{V}_2(\text{PO}_4)_3$ (see **Table 1** of main text), i.e., the electrochemically-ground-state with fully occupied Na(1) (e-GS), metastable- α with Na(1)-occ of 0.75, metastable- β with Na(1)-occ of 0.5 and metastable- γ with Na(1)-occ of 0.25.

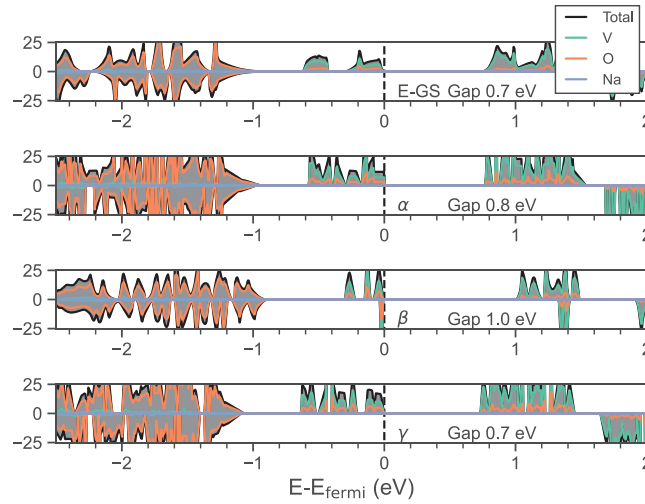


Figure S6. Computed DOS of different phases $\text{Na}_1\text{V}_2(\text{PO}_4)_3$ projected on vanadium, oxygen, and sodium atoms.

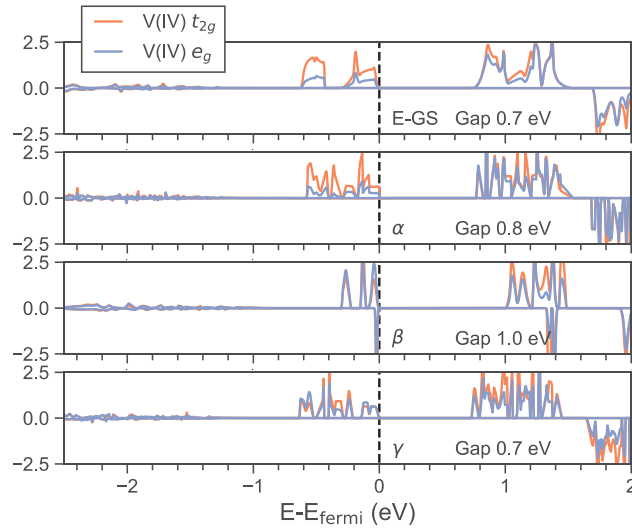


Figure S7. The computed DOS for different phases of $\text{Na}_1\text{V}_2(\text{PO}_4)_3$ projected on $3d$ orbitals (i.e., t_g and e_g) of vanadium. Note that all vanadium sites within the 4 phases of $\text{Na}_1\text{V}_2(\text{PO}_4)_3$ exhibit +4 oxidation state (i.e., V(IV)).

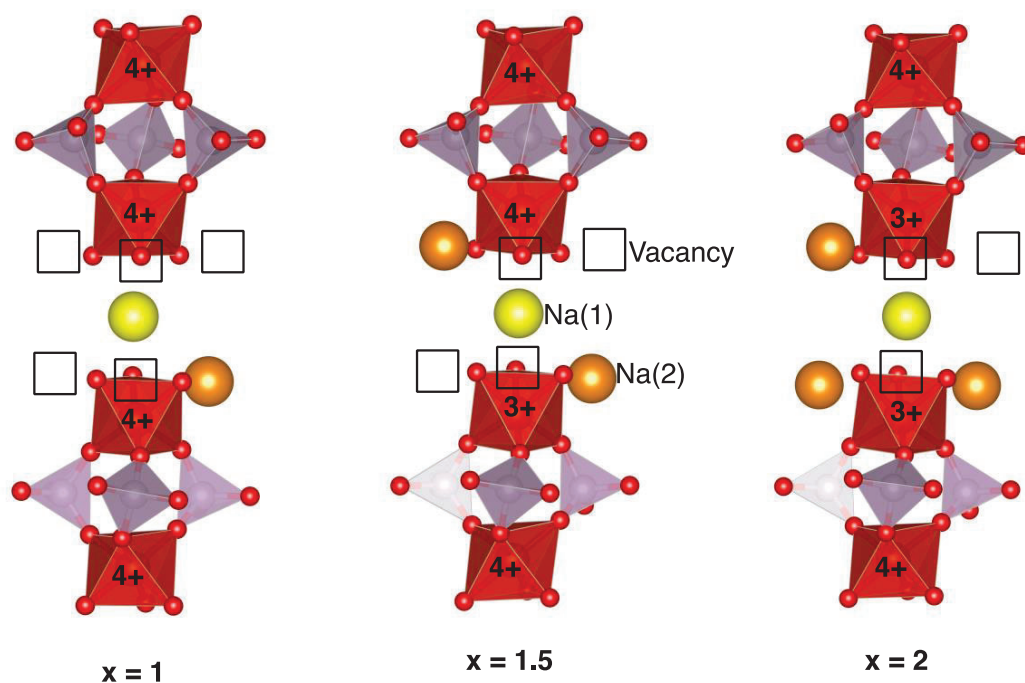


Figure S8. The local Na vacancy and vanadium charge environment of metastable phases of $\text{Na}_x\text{V}_2(\text{PO}_4)_3$, i.e., at $x = 1$, $x = 1.5$, and $x = 2$, corresponding to the blue phases in **Table 1** of the main manuscript. Notably, the computationally constructed structures are in the size of 2 formula units, i.e., $\text{Na}_{2x}\text{V}_4(\text{PO}_4)_6$. All the $\text{V}_2(\text{PO}_4)_3$ “lantern units” are shown. Purple tetrahedra are $(\text{PO}_4)^{3-}$ polyanion groups. Charge states of vanadium sites (red octahedra) have been denoted as 3+ and 4+ for V^{3+} and V^{4+} , respectively.

The metastable phases in **Table 1** of main manuscript correspond to the computationally-predicted phases that exhibit similar Na(1)/Na(2) occupation factors with the experimentally-synthesized phases. As shown in **Figure S6**, the Na(1) site is half occupied (Na1-occ = 0.5) in all the three metastable phases. Furthermore, each Na(1) site is surrounded by 1, 2, and 3 Na(2) ions at Na compositions such as $x = 1$, $x = 1.5$, and $x = 2$, respectively. In addition, the occupied Na(1) sites are surrounded by vanadium sites with lower charge states. For example, at $x = 1.5$, the distance of Na(1)- V^{3+} is $\sim 3.07 \text{ \AA}$, which is shorter than $\sim 3.12 \text{ \AA}$ for Na(1)- V^{4+} . At $x = 2$, the occupied Na(1) sites are closer to V^{3+} than V^{4+} , whose charges are marked

Table S1. Comparison between Rietveld refinements obtained for chemically prepared $\text{Na}_2\text{V}_2(\text{PO}_4)_3$ collected with synchrotron X-ray (black), lab X-ray with normal holder (red), lab X-ray with *in situ* cell (blue), and lab X-ray with *in situ* cell after intensity correction (green). * Note that the occupancy factors of Na(1) and Na(2) sites dramatically changed before and after the intensity correction within in-situ cell.

$\text{Na}_2\text{V}_2(\text{PO}_4)_3$ synchrotron, standard holder , in-situ cell , in-situ cell with intensity correction						
Space group: $R\text{-}3c$ (#167); $Z = 6$						
$a = 8.6599(2) \text{ \AA}$; $a = 8.6612(3) \text{ \AA}$; $a = 8.6506(4) \text{ \AA}$; $a = 8.6497(4) \text{ \AA}$; $c = 21.8882(8) \text{ \AA}$; $c = 21.8903(14) \text{ \AA}$; $c = 21.8864(14) \text{ \AA}$; $c = 21.8865(14) \text{ \AA}$; $c/a = 2.528$; $c/a = 2.527$; $c/a = 2.530$; $c/a = 2.530$; $V = 1421.58(8) \text{ \AA}^3$; $V/Z = 236.929(13) \text{ \AA}^3$ $V = 1422.14(12) \text{ \AA}^3$; $V/Z = 237.02(2) \text{ \AA}^3$ $V = 1418.40(12) \text{ \AA}^3$; $V/Z = 236.40(2) \text{ \AA}^3$ $V = 1418.12(12) \text{ \AA}^3$; $V/Z = 236.35(2) \text{ \AA}^3$						
$R_{\text{wp}} = 15.2 \%$; $R_p = 18.1 \%$; $R_{\text{Bragg}} = 7.46 \%$ $R_{\text{wp}} = 22.2 \%$; $R_p = 18.7 \%$; $R_{\text{Bragg}} = 7.33 \%$ $R_{\text{wp}} = 18.5 \%$; $R_p = 34.0 \%$; $R_{\text{Bragg}} = 8.41 \%$ $R_{\text{wp}} = 19.9 \%$; $R_p = 46.6 \%$; $R_{\text{Bragg}} = 2.38 \%$						
Atom	Wyckoff position	x/a	y/b	z/c	Uiso, $\text{\AA}^2$	Occ.
V(1)	12c	0	0	0.1482(3) 0.1484(7) 0.1430(10) 0.1438(8)	0.012(2) 0.029(7) 0.007* 0.008*	1
P(1)	18e	0.2754(13) 0.283(3) 0.279(3) 0.279(2)	0	0.25	0.066(6) 0.109(17) 0.013* 0.040*	1
Na(1)	6b	0	0	0	0.033(14) 0.001(4) 0.013* 0.025*	0.66(4) 0.64(7) 0.94(5) 0.70(6)
Na(2)	18e	0.667(3) 0.668(5) 0.660(5) 0.643(7)	0	0.25	0.050(17) 0.05(5) 0.013* 0.035*	0.55(3) 0.57(6) 0.77(2) 0.49(2)
O(1)	36f	0.018(2) 0.018(3) 0.019(5) 0.017(6)	0.1944(17) 0.197(3) 0.198(5) 0.186(4)	0.1915(5) 0.1944(10) 0.1886(10) 0.1930(8)	0.023(6) 0.028(15) 0.051* 0.042*	1
O(2)	36f	0.1957(16) 0.193(3) 0.199(4) 0.201(3)	0.1684(16) 0.166(3) 0.175(3) 0.165(3)	0.0834(11) 0.082(2) 0.0686(17) 0.0812(18)	0.060(6) 0.104(13) 0.051* 0.070*	1

Source data tables for Figs. 1–5

Source Data Fig. 1b: X Ray powder patterns of $\text{Na}_3\text{V}_2(\text{PO}_4)_3$, $\text{Na}_1\text{V}_2(\text{PO}_4)_3$, mixtures of $\text{Na}_3\text{V}_2(\text{PO}_4)_3$ & $\text{Na}_1\text{V}_2(\text{PO}_4)_3$ at 30°C/325°C (heating)/500°C (heating)/250°C (cooling) / 35°C cooling

Source Data Fig. 1c: X Ray powder patterns of mixtures of $y \text{Na}_3\text{V}_2(\text{PO}_4)_3 + (1-y) \text{Na}_1\text{V}_2(\text{PO}_4)_3$ before annealing for $y = 0 / 0.25 / 0.375 / 0.5 / 0.625 / 0.75 / 1$

Source Data Fig. 1d: X Ray powder patterns of mixtures of $y \text{Na}_3\text{V}_2(\text{PO}_4)_3 + (1-y) \text{Na}_1\text{V}_2(\text{PO}_4)_3$ after annealing : $\text{Na}_1\text{V}_2(\text{PO}_4)_3 / \text{Na}_{1.5}\text{V}_2(\text{PO}_4)_3 / \text{Na}_{1.75}\text{V}_2(\text{PO}_4)_3 / \text{Na}_2\text{V}_2(\text{PO}_4)_3 / \text{Na}_{2.25}\text{V}_2(\text{PO}_4)_3 / \text{Na}_{2.5}\text{V}_2(\text{PO}_4)_3 / \text{Na}_3\text{V}_2(\text{PO}_4)_3$

Source Data Fig. 2a: X Ray powder pattern of e- $\text{Na}_2\text{V}_2(\text{PO}_4)_3$

Source Data Fig. 3a: Galvanostatic curve of e- $\text{Na}_2\text{V}_2(\text{PO}_4)_3$ & c- $\text{Na}_2\text{V}_2(\text{PO}_4)_3$ between 2.5 and 4.25 V vs Na^+/Na

Source Data Fig. 3d: GITT curve of c- $\text{Na}_2\text{V}_2(\text{PO}_4)_3$

Source Data Fig. 3f: Galvanostatic curve of c- $\text{Na}_2\text{V}_2(\text{PO}_4)_3$ between 3 and 0.5 V vs Na^+/Na

Source Data Fig. 4a: X ray powder patterns from operando measurements in the 2.5–4.4 V region

Source Data Fig. 4b: X ray powder patterns from operando measurements in the 1.3–3 V region

Source Data Fig. 5: X ray diffraction patterns of $\text{V}_2(\text{PO}_4)_3$ chemically deintercalated and $\text{Na}_{0.5}\text{V}_2(\text{PO}_4)_3$ charged at 4.4 V